

Structure and properties of glasses and melts in the $\text{MgO-Al}_2\text{O}_3\text{-SiO}_2$, $\text{CaO-Al}_2\text{O}_3\text{-SiO}_2$, $\text{MgO-CaO-Al}_2\text{O}_3\text{-SiO}_2$ systems

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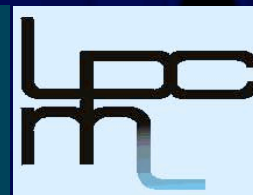
² IMPMC, Universités PARIS 6 et 7, CNRS, 4 place Jussieu, 75252 Paris

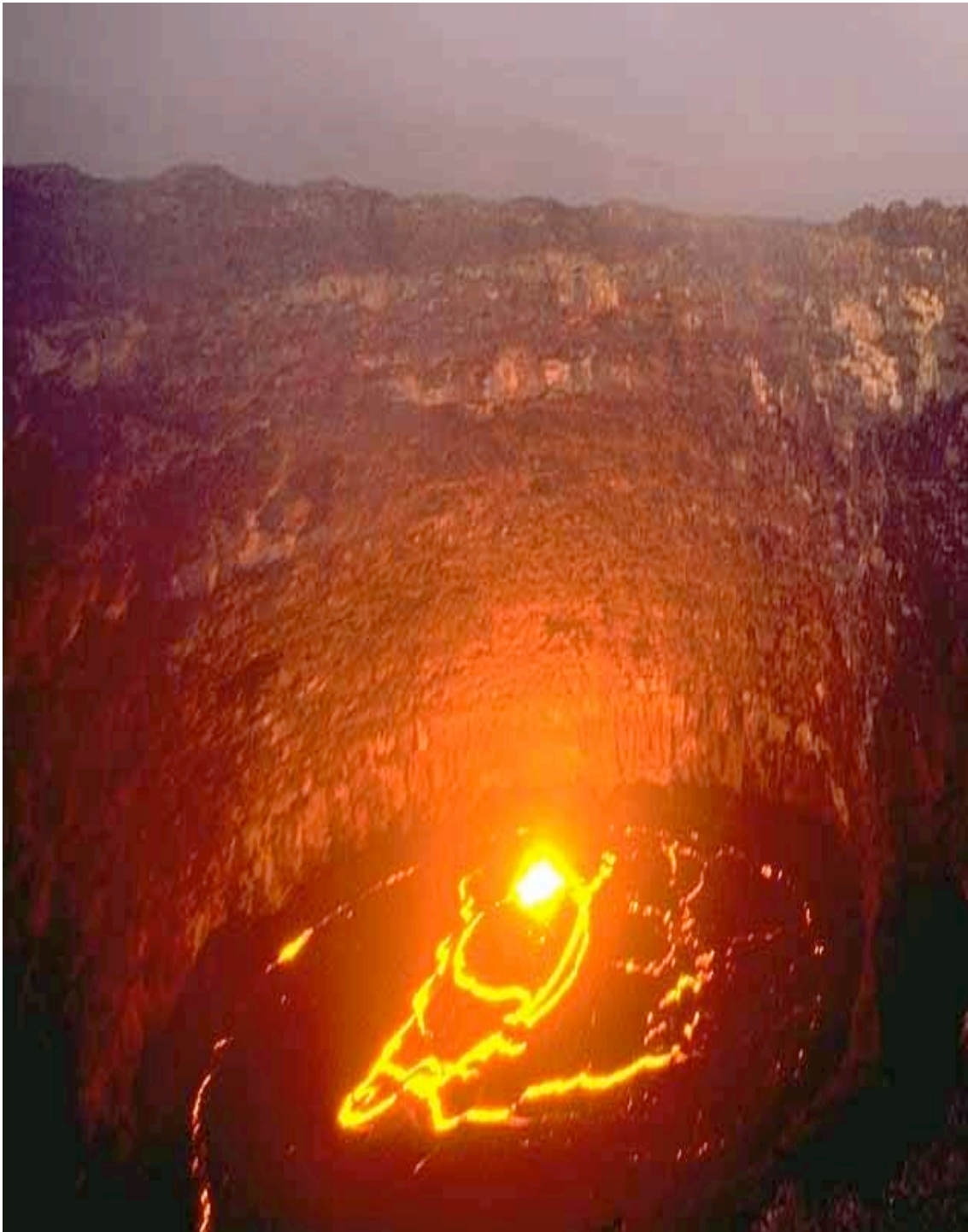
³ LPCML, Université Lyon 1, CNRS UMR 5620, 12 rue Ampère, 69622 Villeurbanne

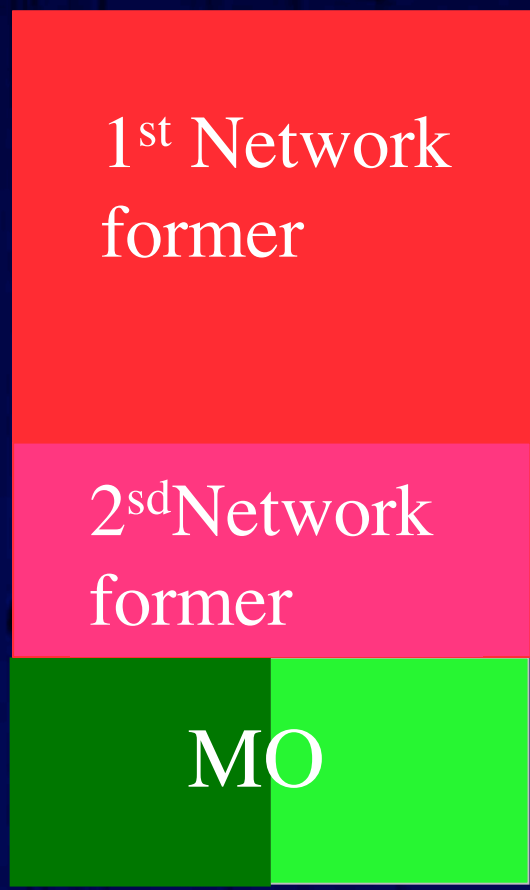
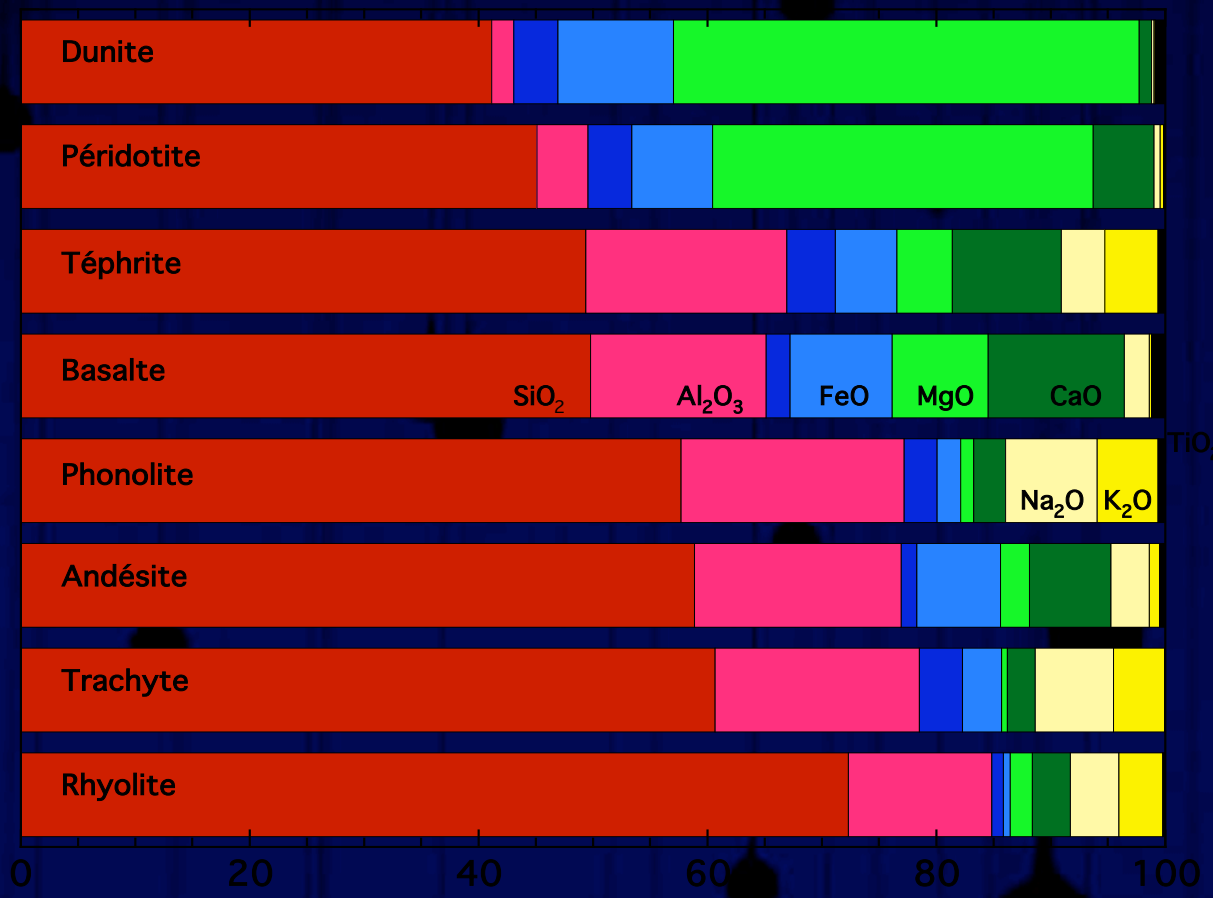
⁴ SOLEIL L'Orme des Merisiers, BP48, 91192 Gif s/Yvette, France.

⁵ Dept of Geology, University of Toronto, 22 Russell St, Toronto, Canada

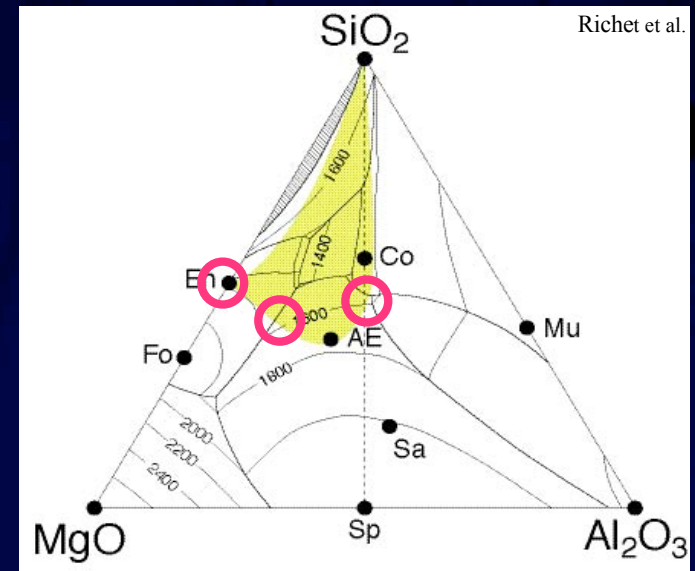
⁶ CRMHT-CNRS, 1D av. Recherche Scientifique, 45071 Orléans cedex 2



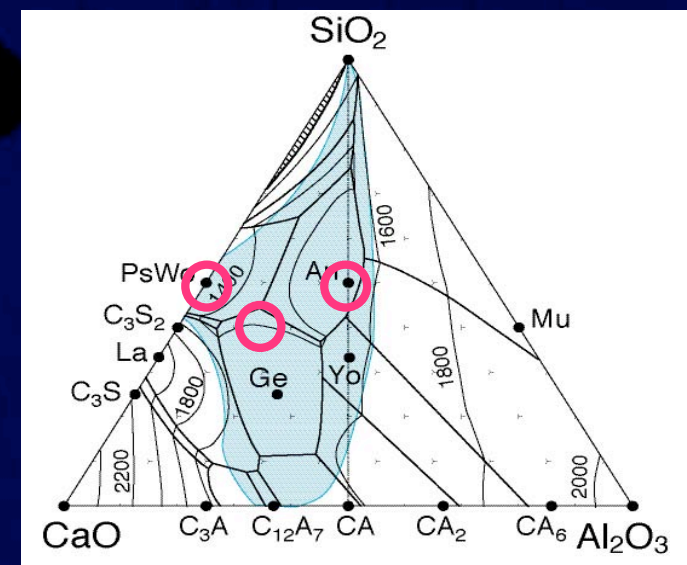




1) MAS System

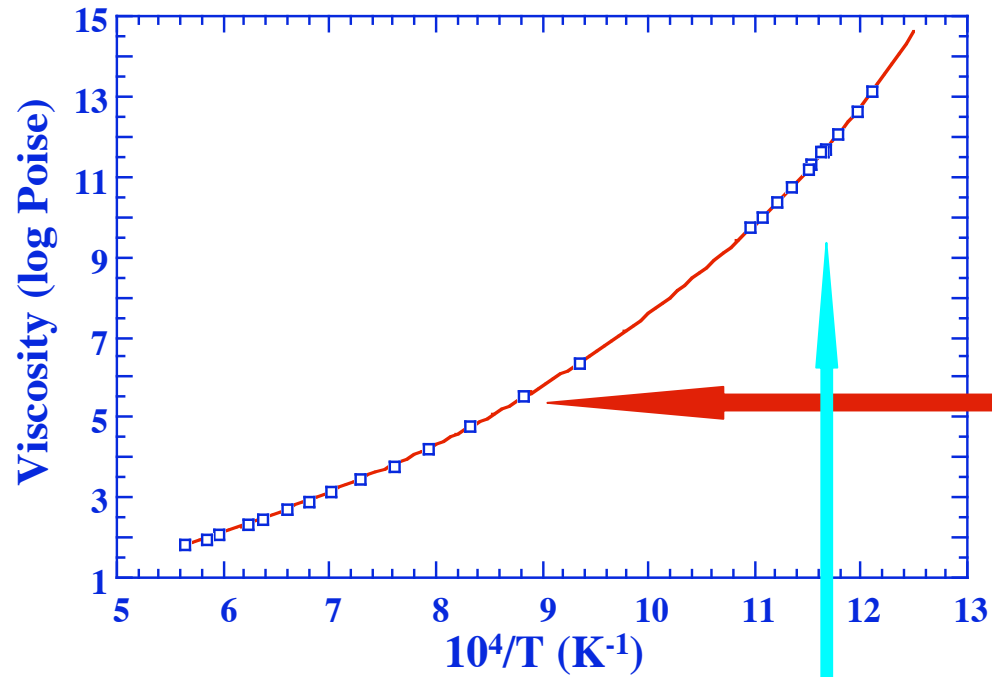


3) CMAS System

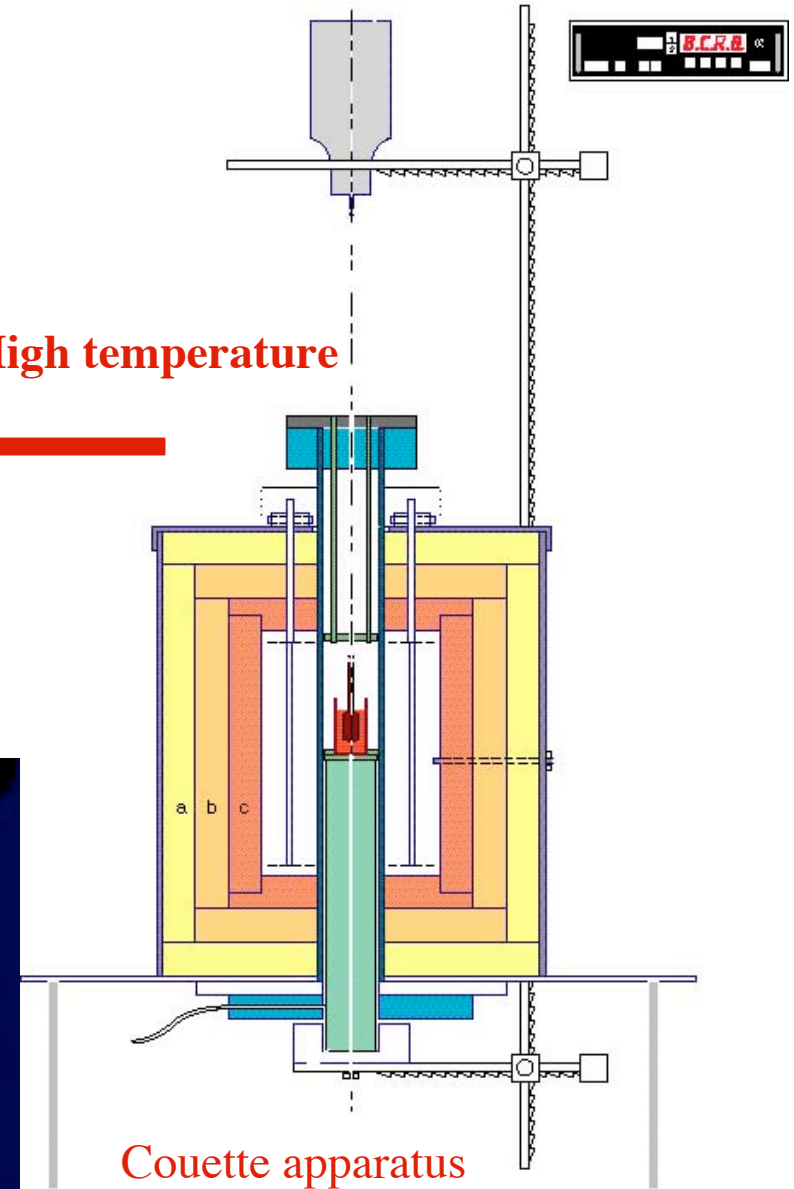


2) CAS System

Viscosity measurements



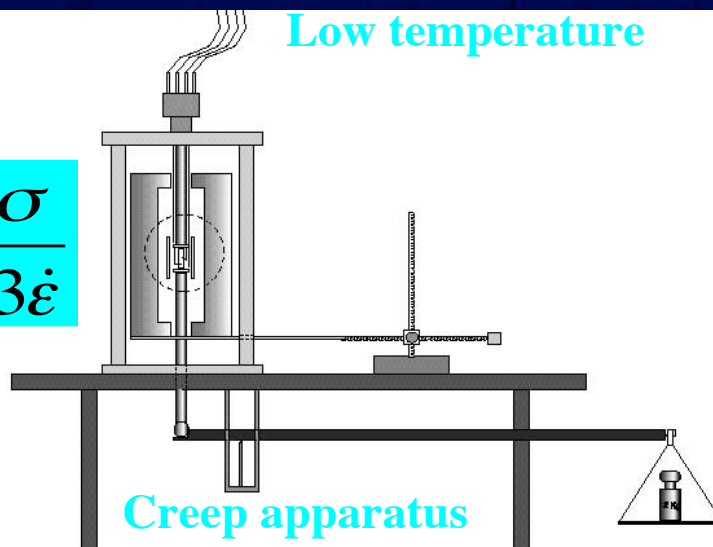
High temperature



Couette apparatus

$$\eta = \frac{\sigma}{3\dot{\epsilon}}$$

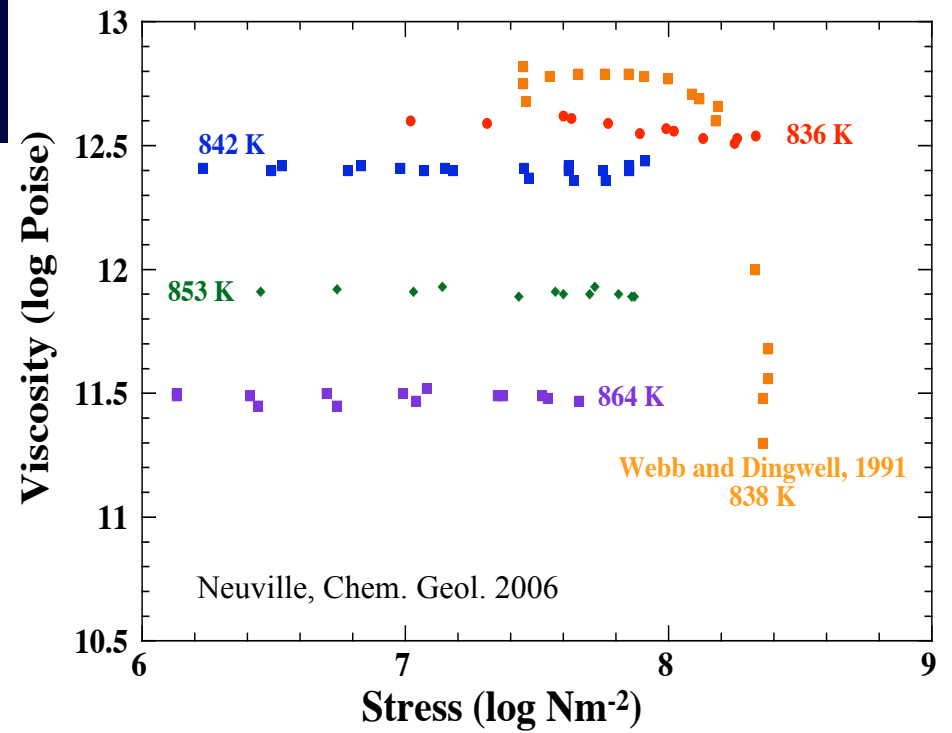
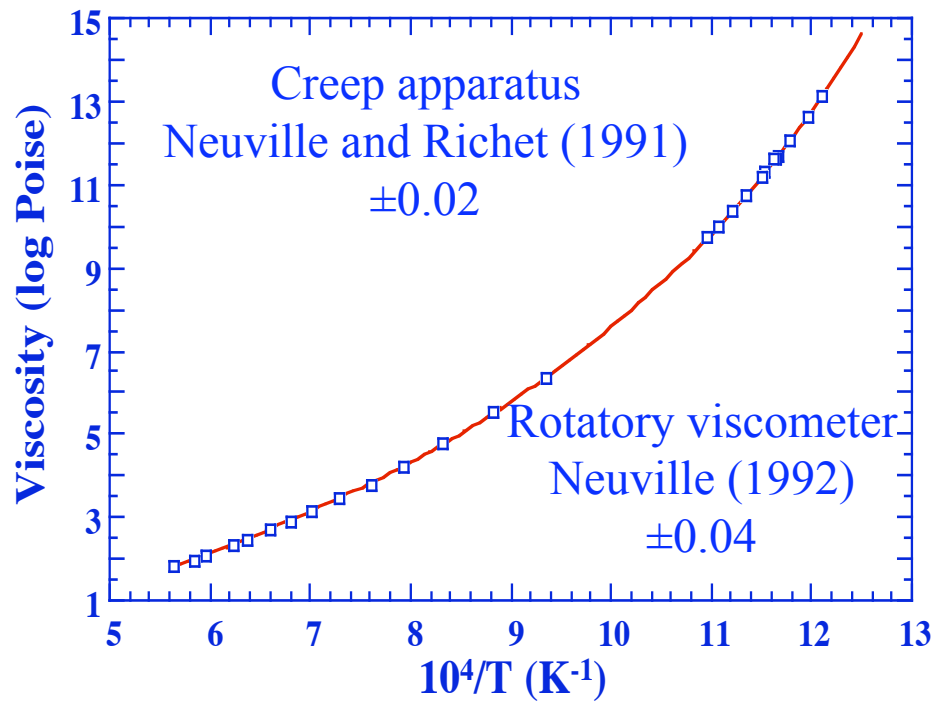
Low temperature



Creep apparatus

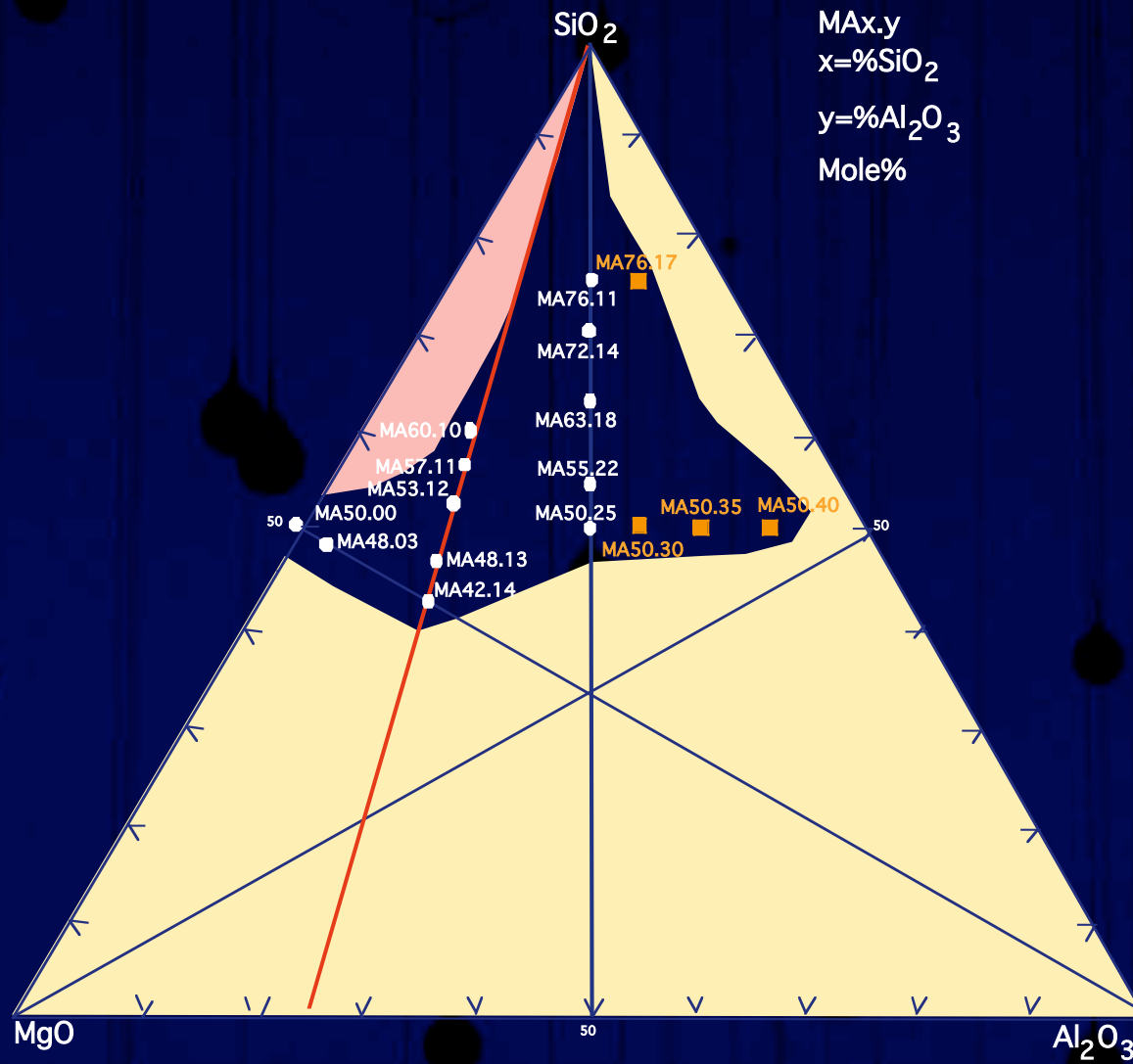
Viscosity measurements

NBS710



$$T_g \Rightarrow \eta = 10^{13} \text{ Poise}$$

MAS



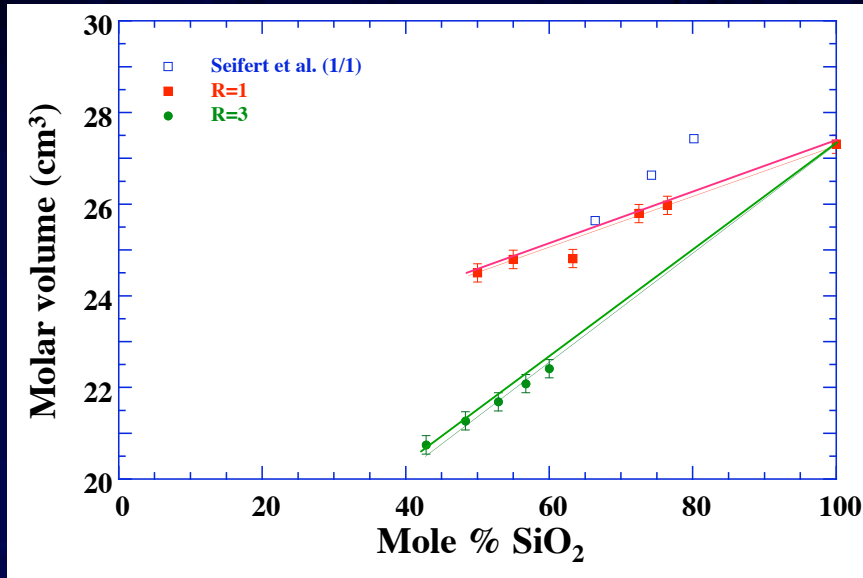
MA_{x.y}
x=%SiO₂
y=%Al₂O₃
Mole%

Normal quench

Rapid quench



MV of glass increases linearly with SiO_2 and decreases with MgO

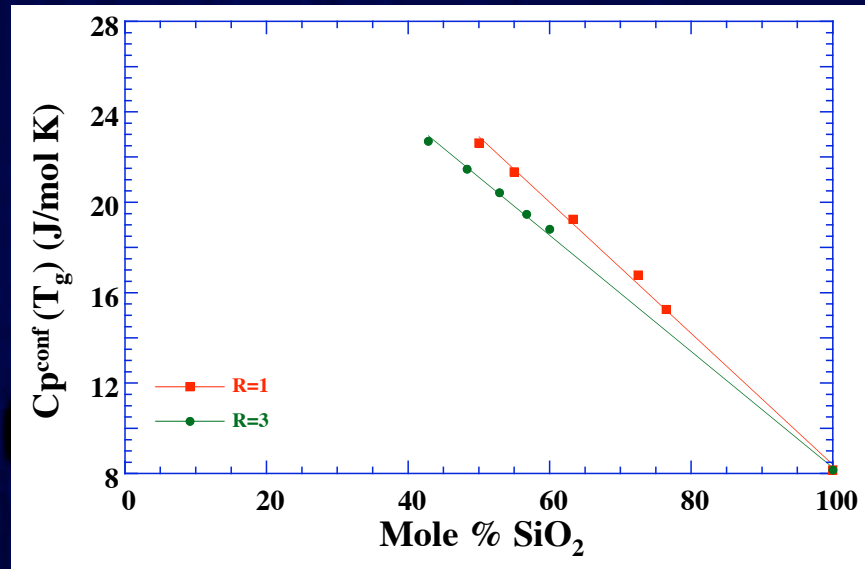


$$C_p^{\text{conf}} = C_p^1 - C_{pg}(T_g)$$

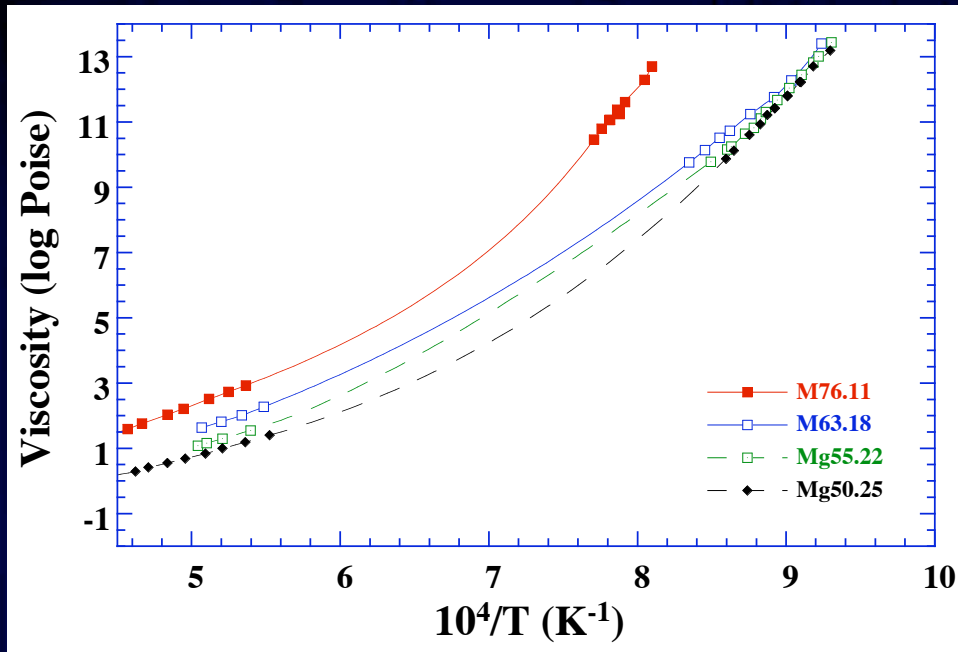
$$T_g \Rightarrow \log \eta = 13 \log P_0$$

Configurational heat capacity increases with decreasing SiO_2 and increasing Al_2O_3 content.

After Richet (1987)
and Richet and Bottinga (1984)

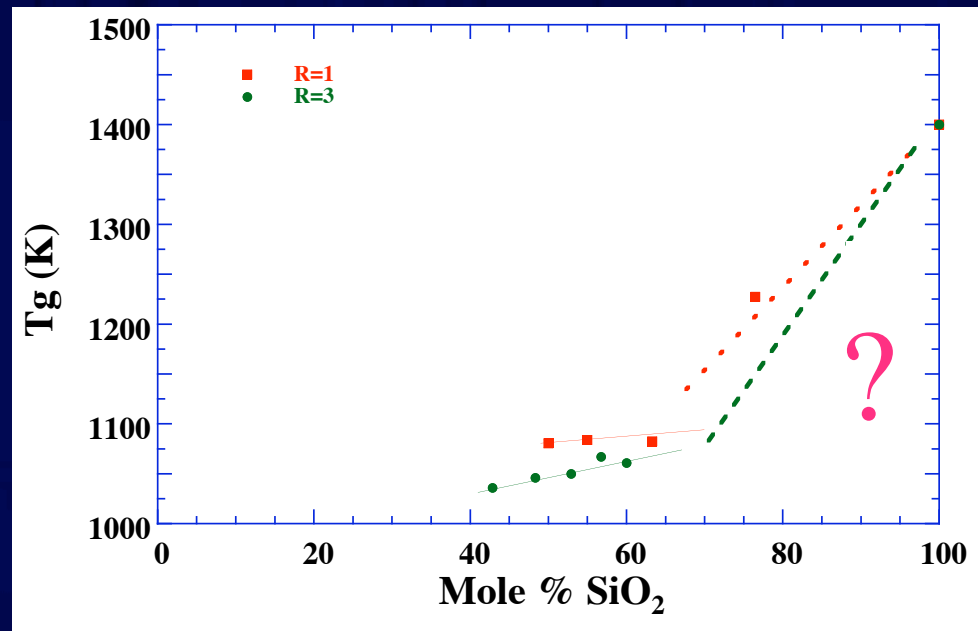


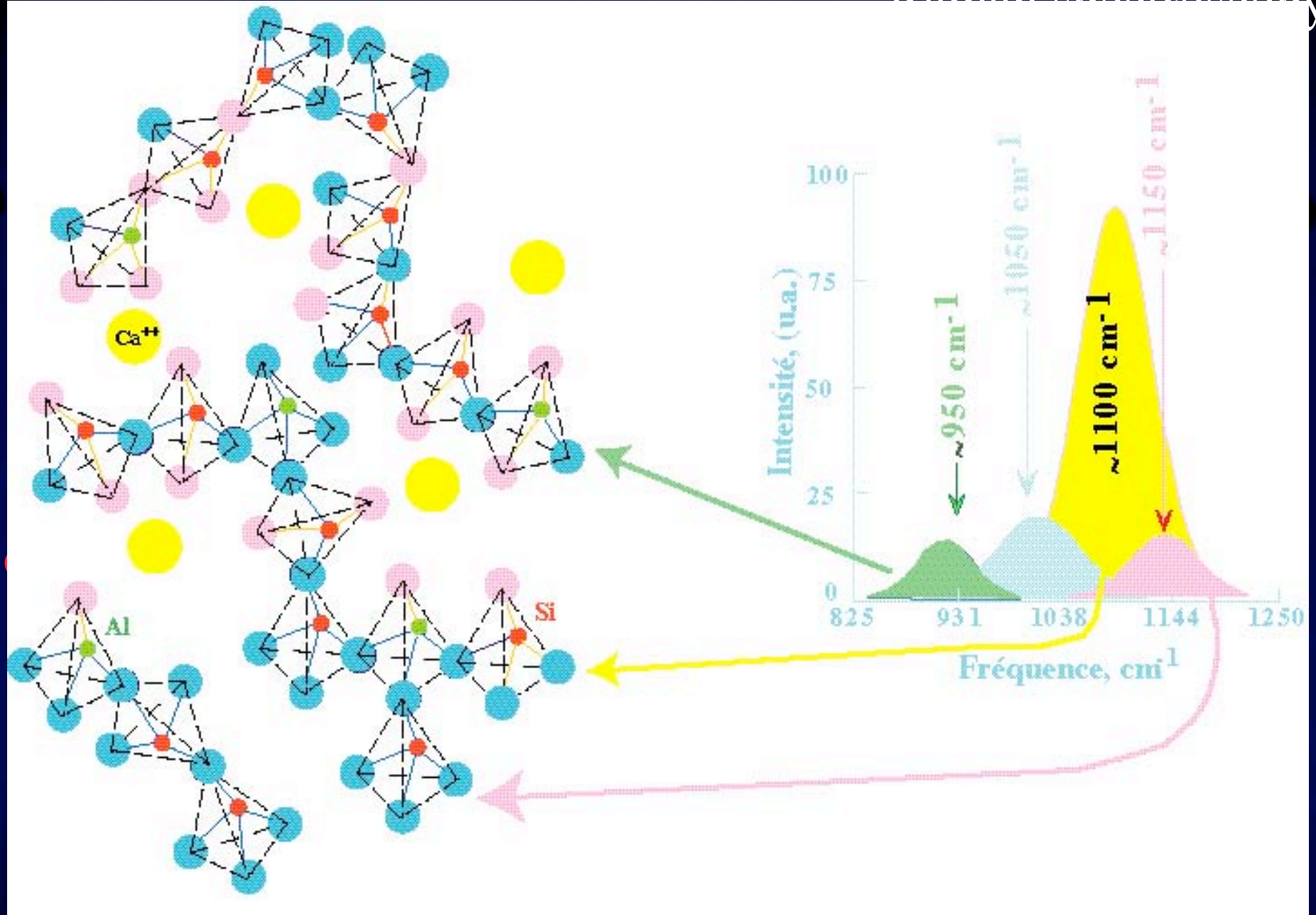
Viscosity MAS



Viscosity increases strongly with SiO_2

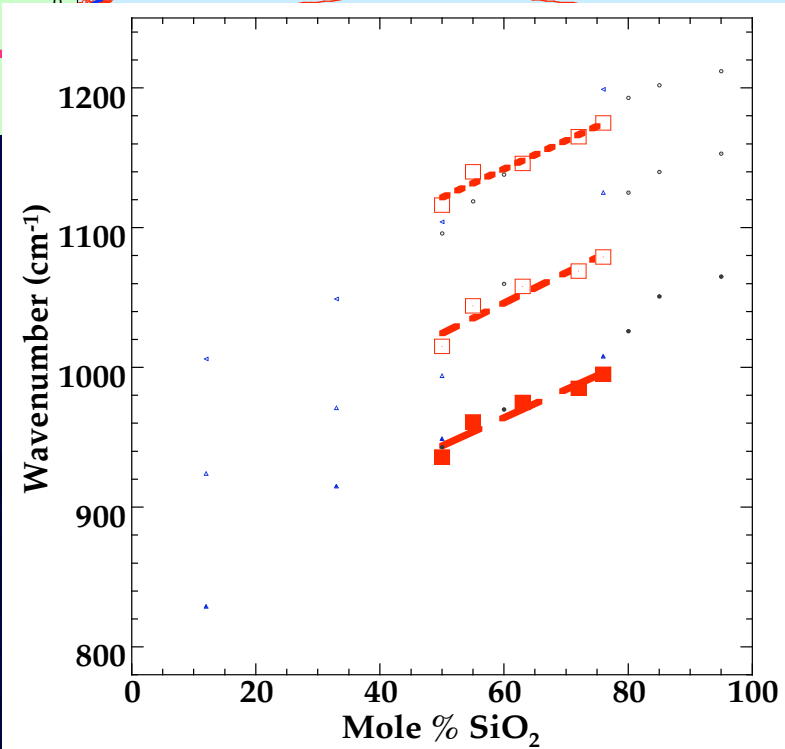
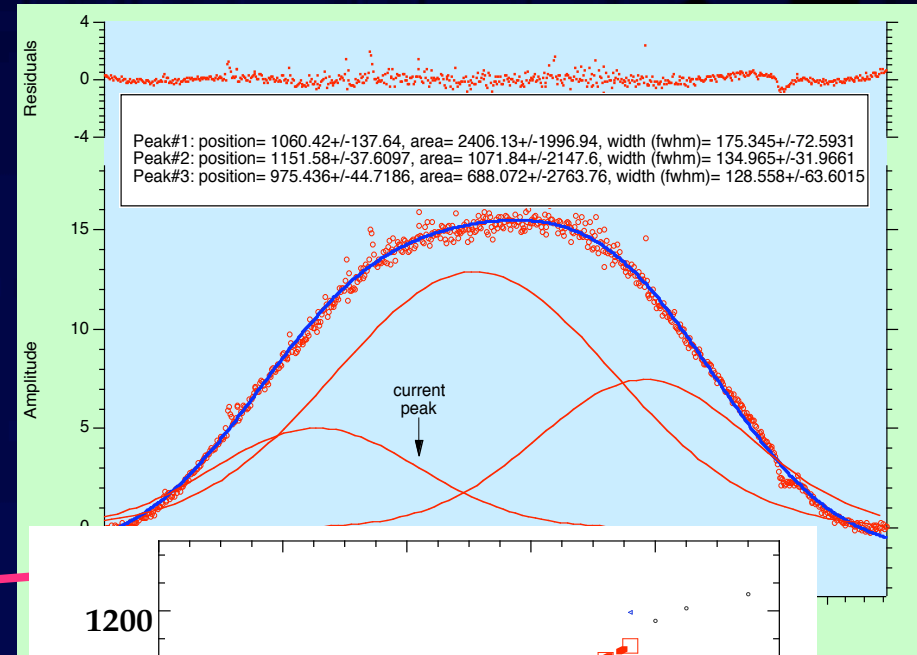
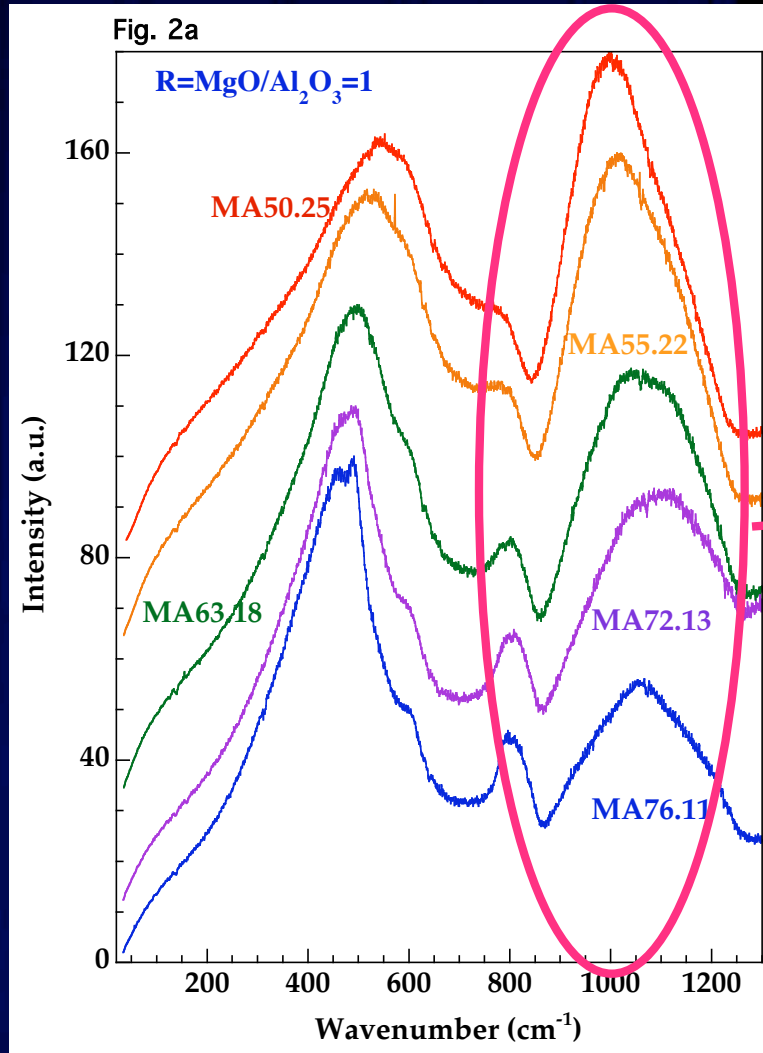
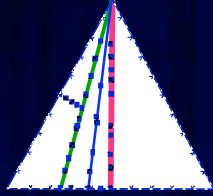
T_g increases strongly with SiO_2 and very small variation between 50 and 70 mole % of SiO_2

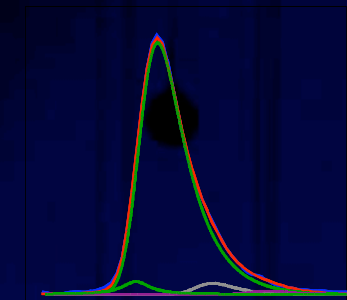




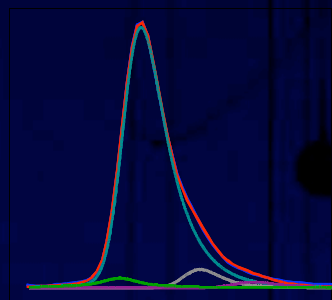
n: number of bridging oxygen

Raman spectroscopy MAS

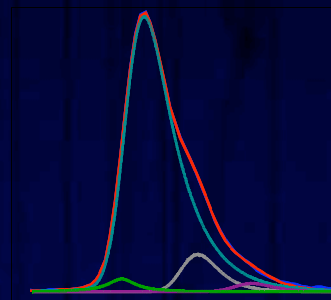




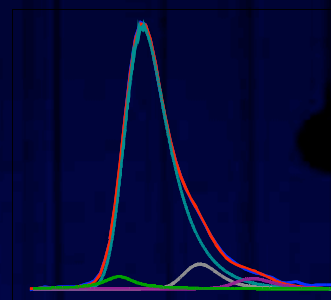
MA60.10



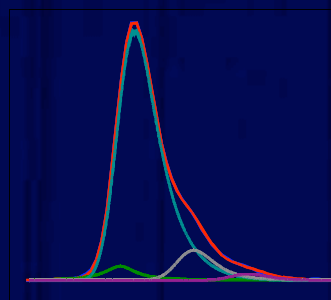
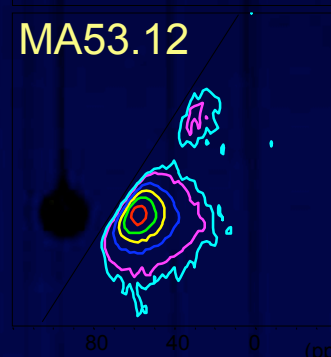
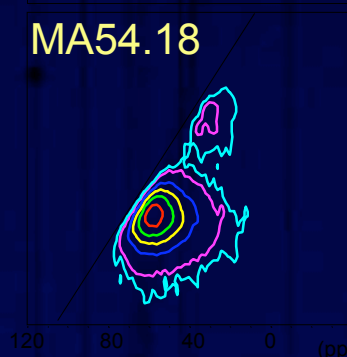
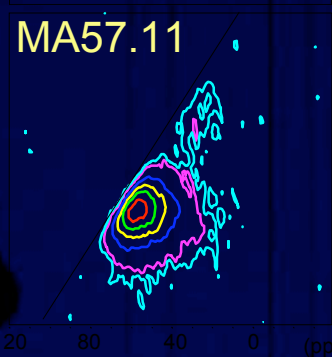
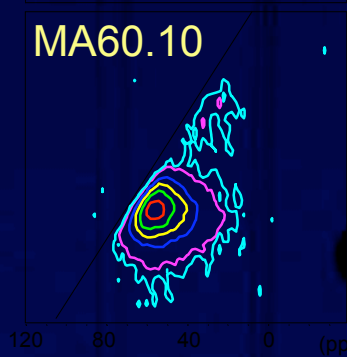
MA57.11



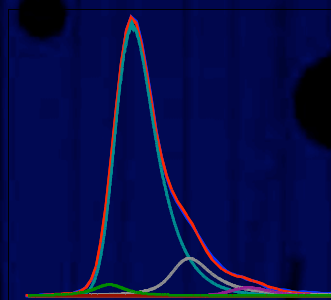
MA54.18



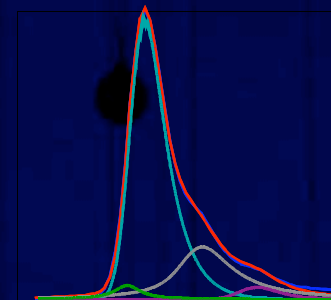
MA53.12



MA48.13

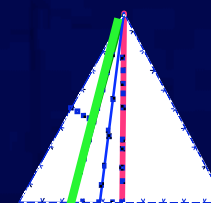


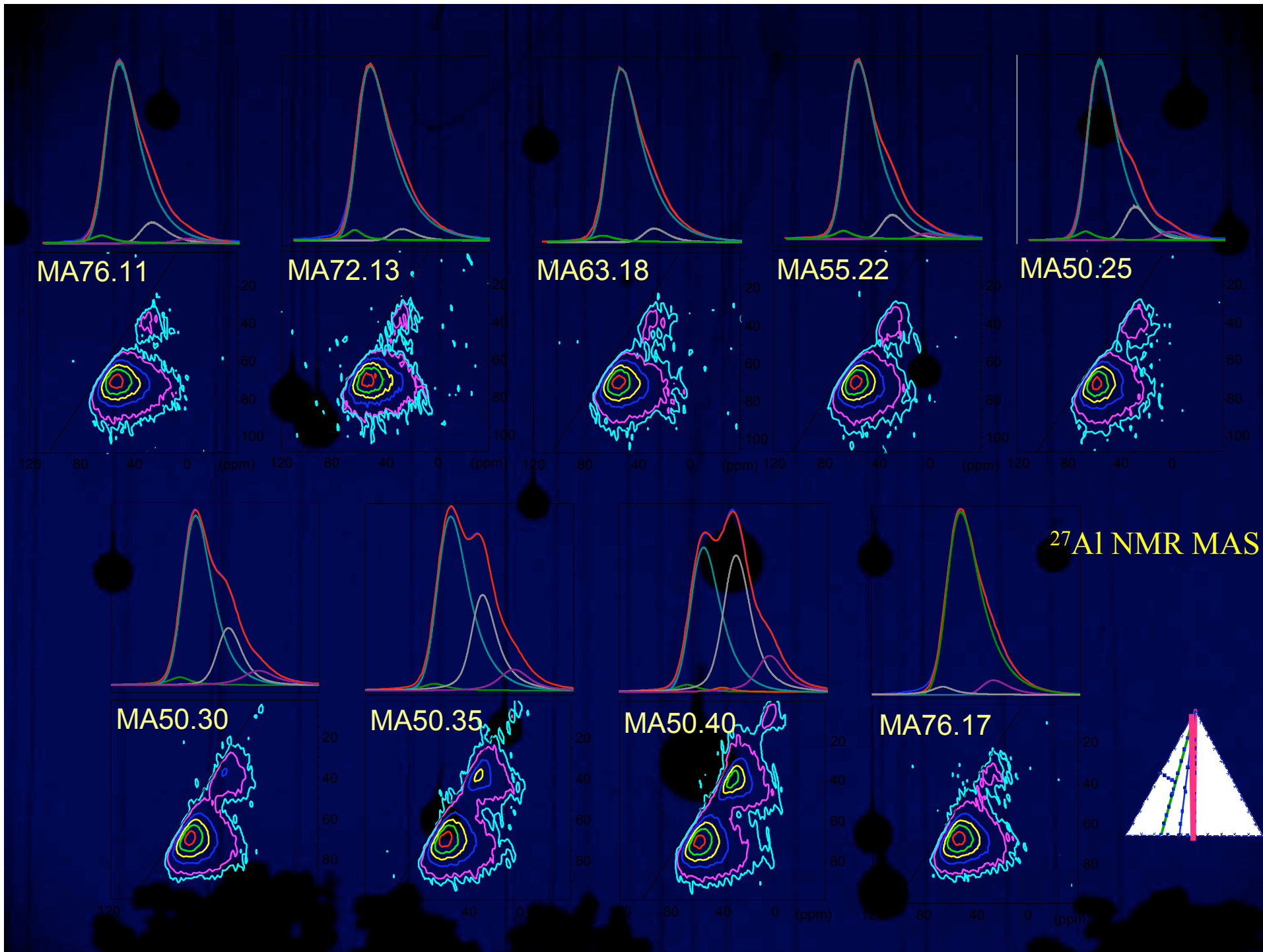
MA42.14



MA48.03

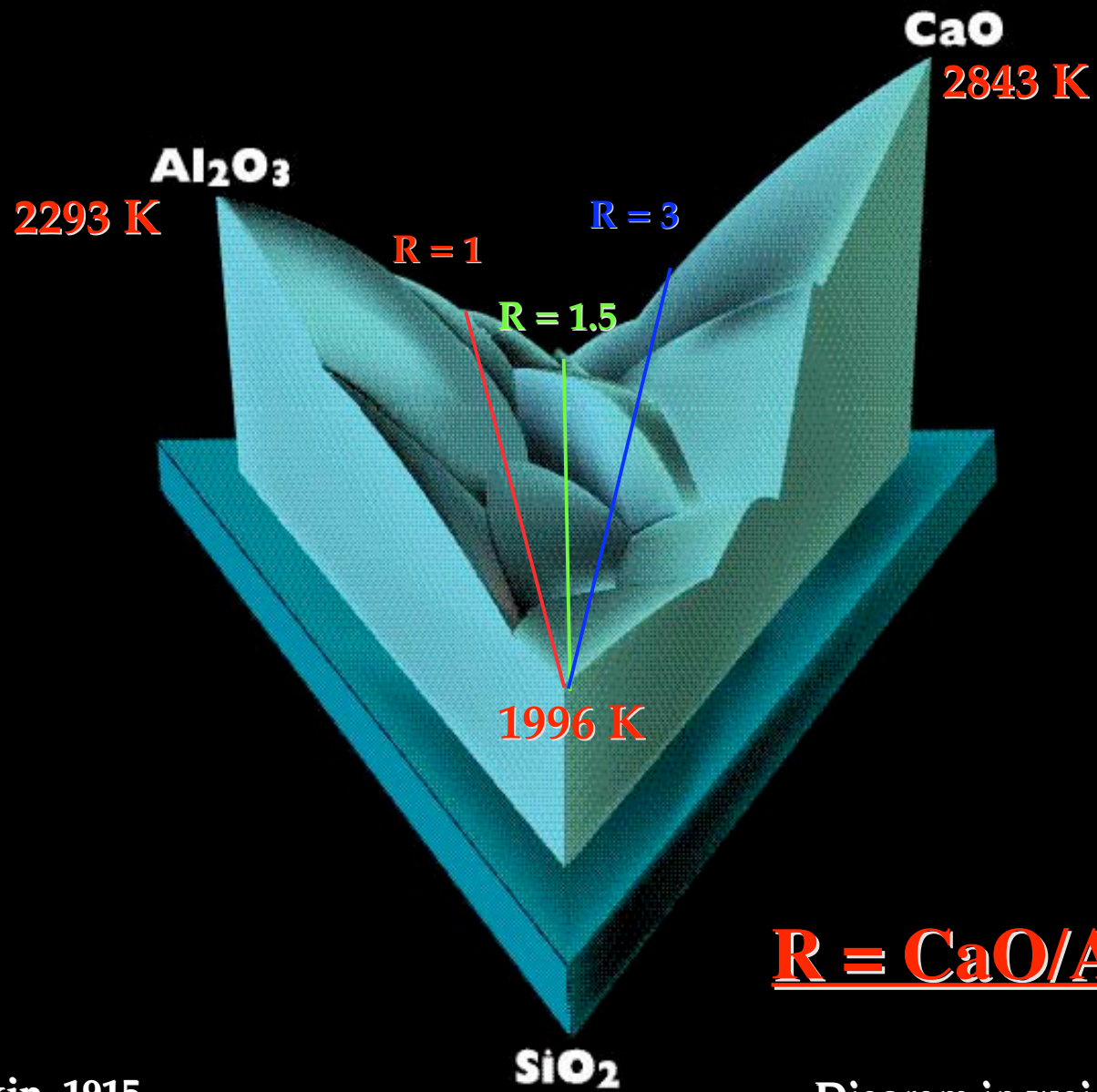
²⁷Al NMR MAS





Mg - Conclusions

- **MV increases and C_p^{conf} decreases with SiO_2**
- **Viscosity and T_g increase with SiO_2 and Al_2O_3**
- **Viscosity and Raman spectroscopy for $R=1$ glasses => random substitution Si/Al, few structural changes**
- **NMR => ^VAl increases with Al_2O_3**



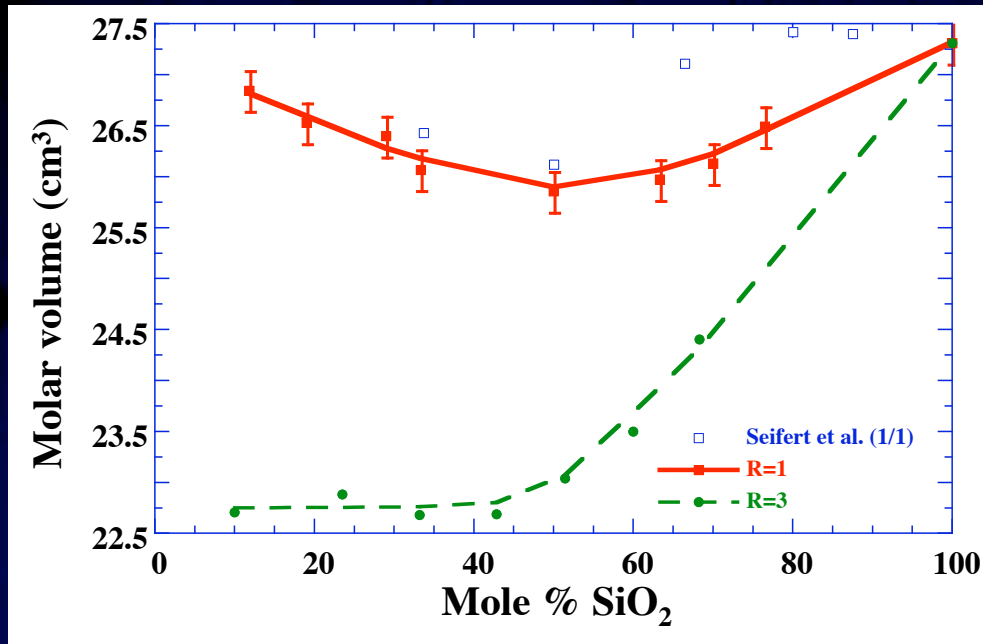
From Rankin, 1915

$$R = \frac{\text{CaO}}{\text{Al}_2\text{O}_3}$$

Diagram in weight %

CAS

MV of glass increases
with SiO_2
and decreases with CaO

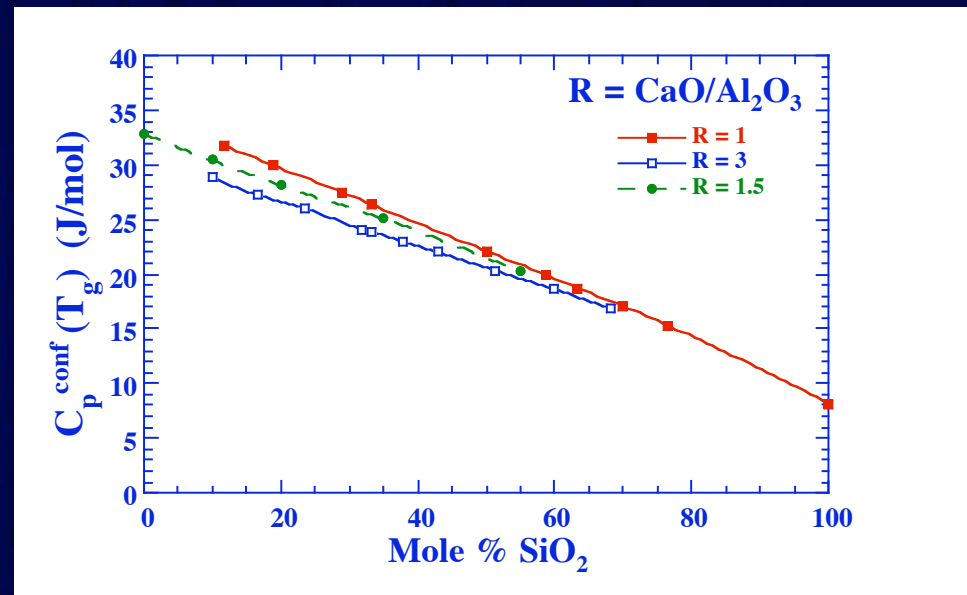


$$C_p^{\text{conf}} = C_p^1 - C_{pg}(T_g)$$

$$T_g \Rightarrow \log \eta = 13 \log P_0$$

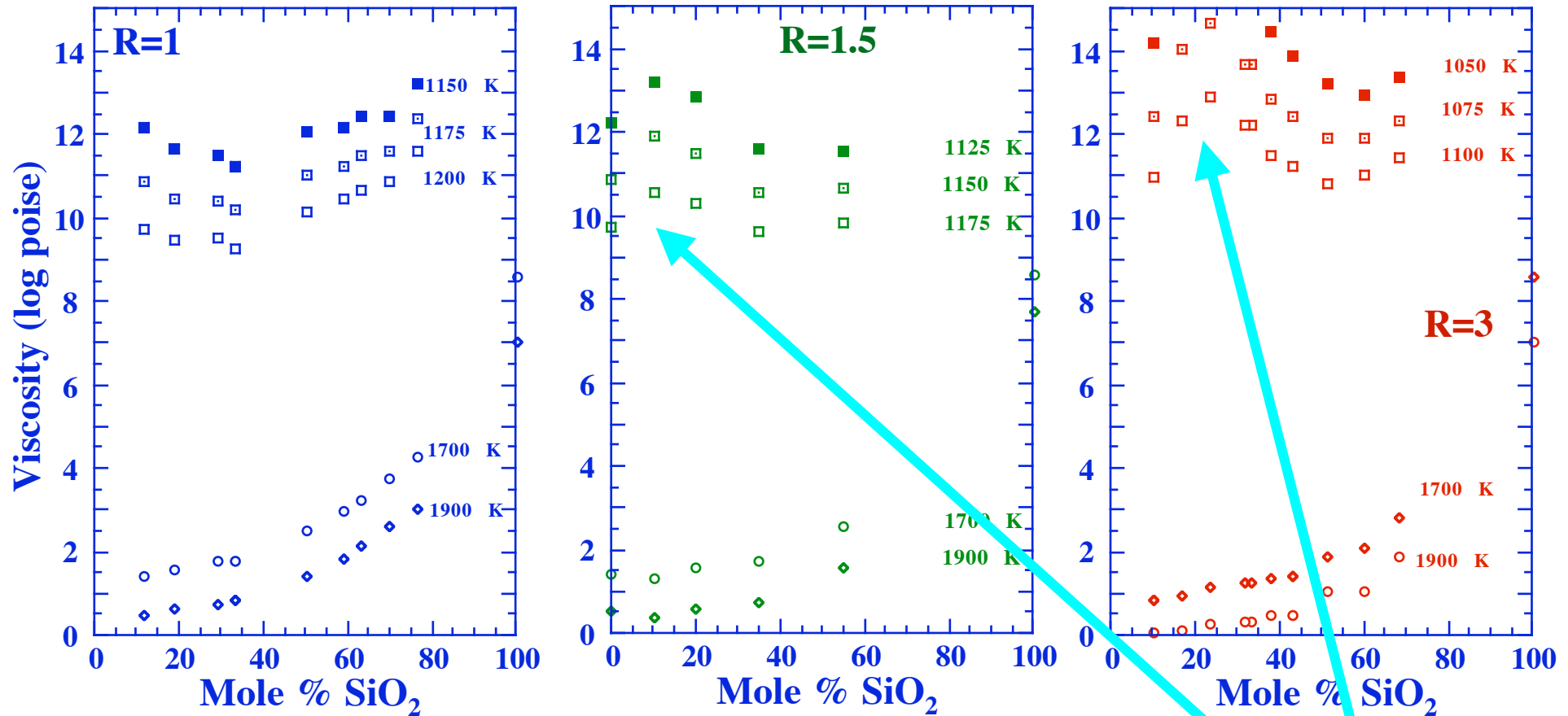
Configurational heat capacity
increases with decreasing SiO_2
and increasing Al_2O_3 content.

After Richet (1987)
and Richet and Bottinga (1984)



Viscosity as a function of SiO₂ for 3 joins 1 < R = CaO/Al₂O₃ < 3

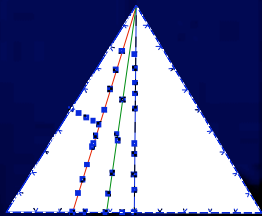
CAS



Viscosity decreases with decreasing SiO₂ and increasing CaO

Decrease of viscosity from R=1 to R=3

Increase of viscosity at low SiO₂



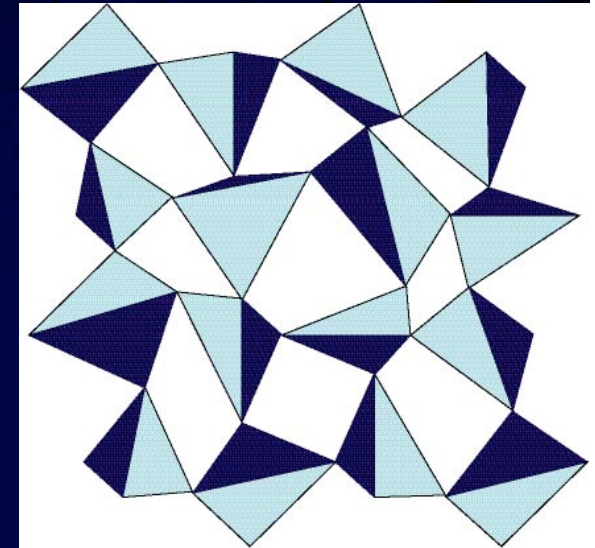
Neuville, 1992

Glass transition temperature

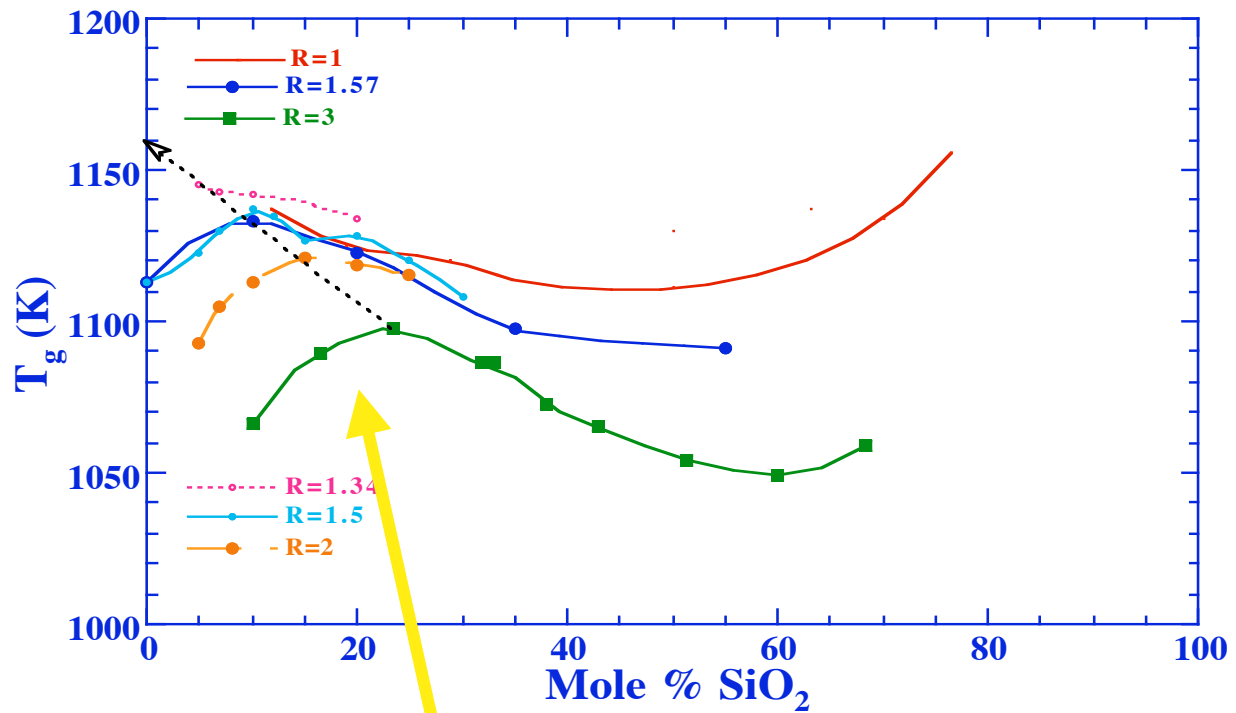
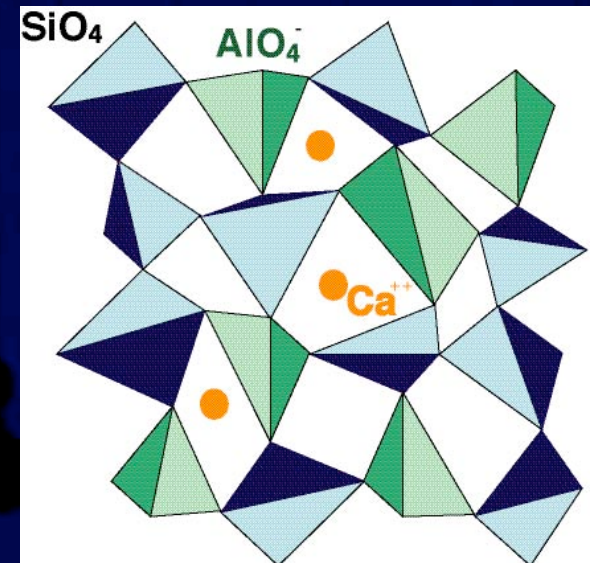
CAS

$$T_g \Rightarrow \eta = 10^{13} \text{ poise}$$

$T_g = 1400 \text{ K}$
SiO₂ glass



T_g decrease
with SiO₂

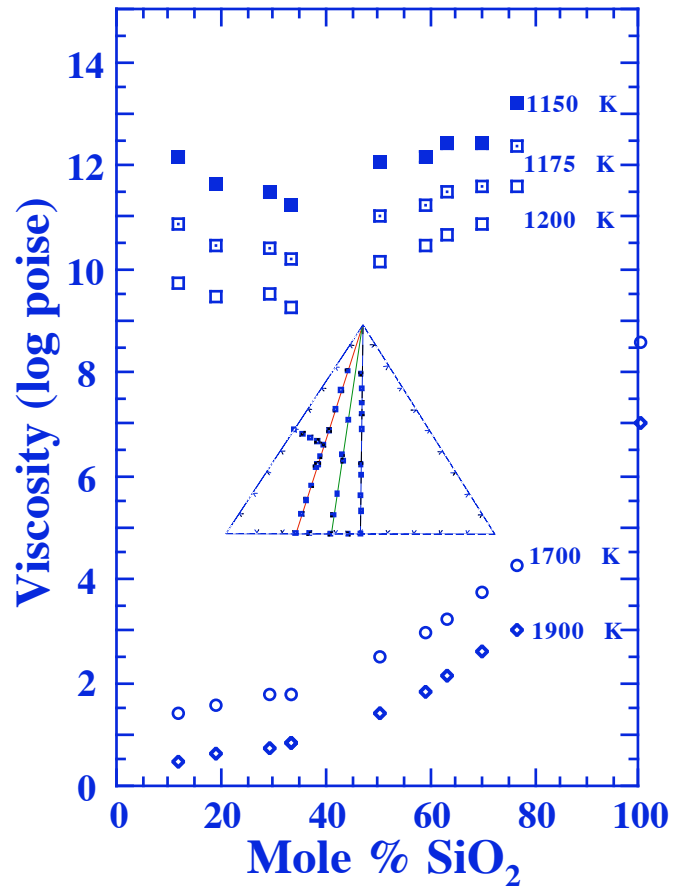


maximum in T_g at 10-20 mol% SiO₂

Configuration Entropy

CAS

R = 1



R = CaO/Al₂O₃

$$\log \eta = A_e + B_e/TS^{conf}(T)$$

$$S^{conf}(T_g) = S^{mix} + \sum x_i S_i^{conf}(T_g)$$

$$S^{conf}_{top} = \sum x_i S_i^{conf}(T_g)$$

$$S^{mix} = -nR \sum X_i \ln X_i$$

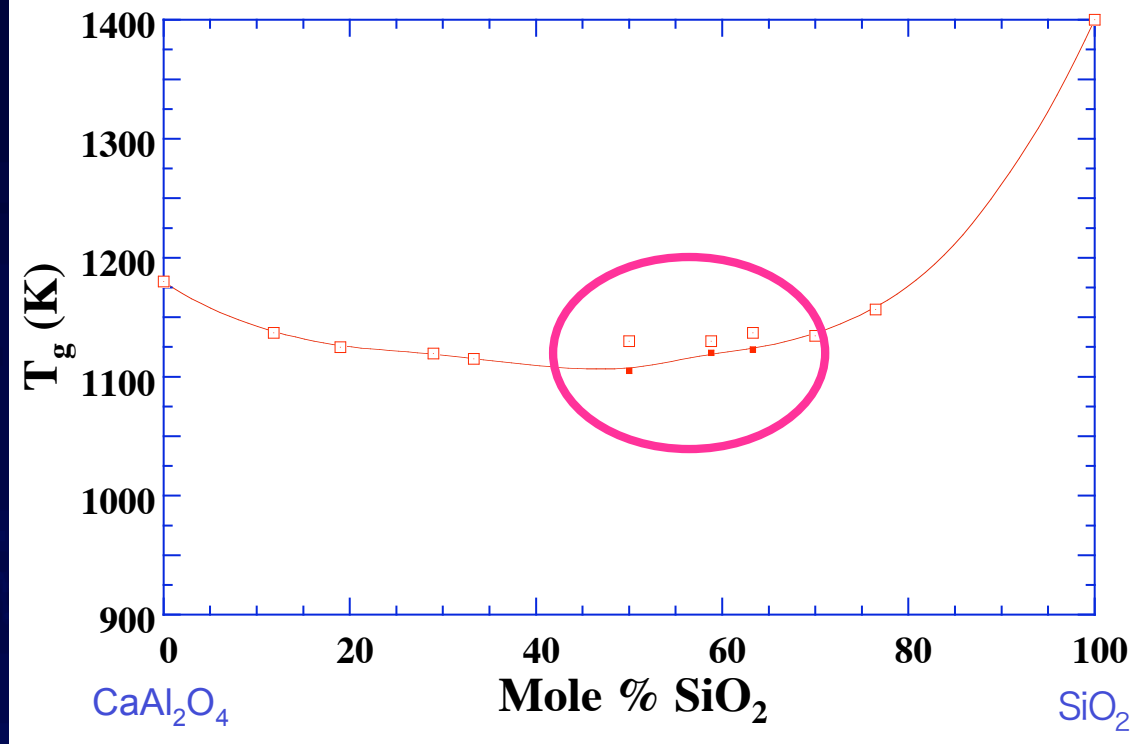
$$X_i = Al/(Al+Si)$$

Ideal mixing => random distribution

$$S^{conf}(T) = S^{conf}(T_g) + \int_{T_g}^T C_p^{conf}/T dT$$

$$with C_p^{conf} = C_p^l - C_{pg}(T_g)$$

Substitution of Si by Al
in Q⁴ species along the join R=1

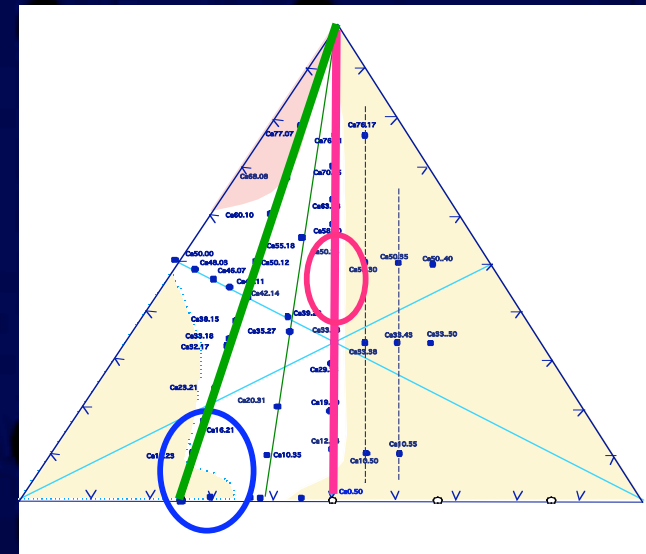


CAS

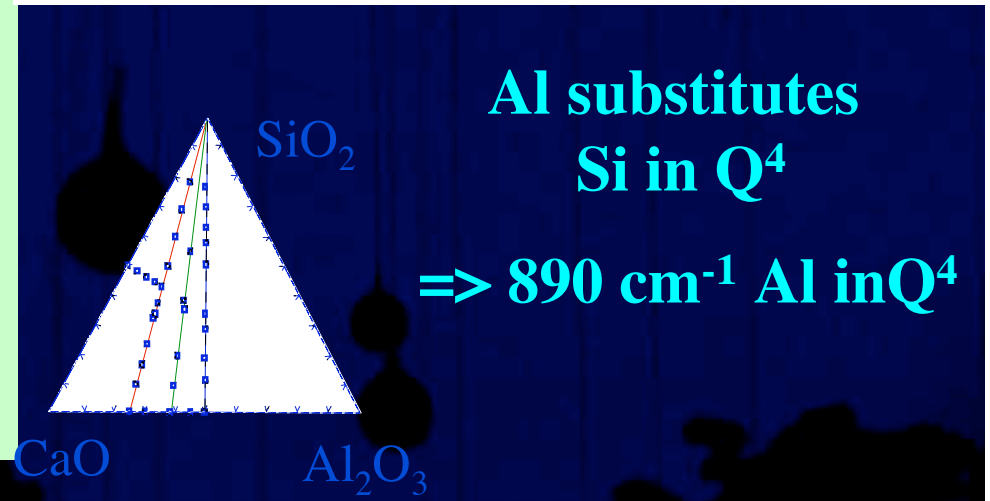
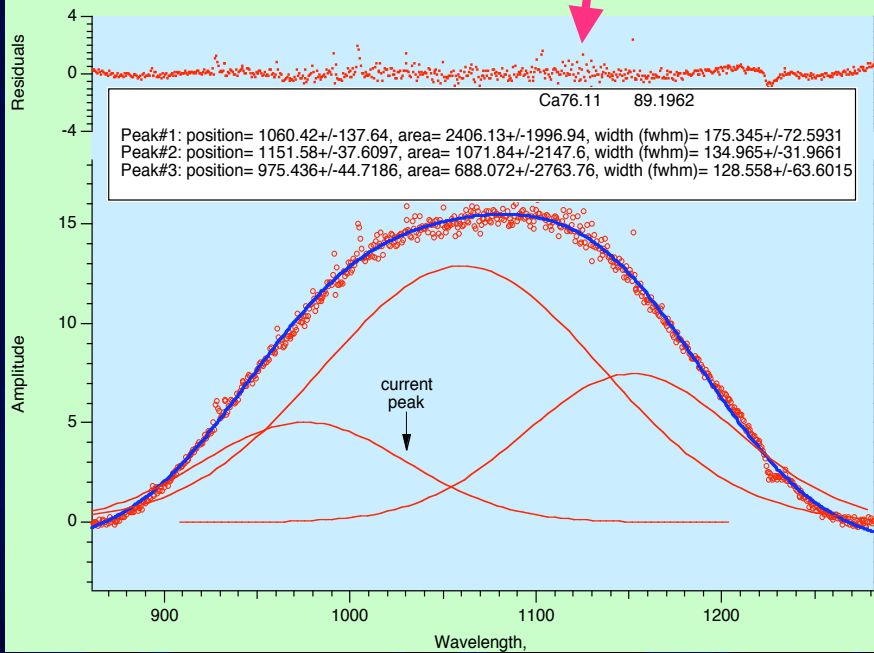
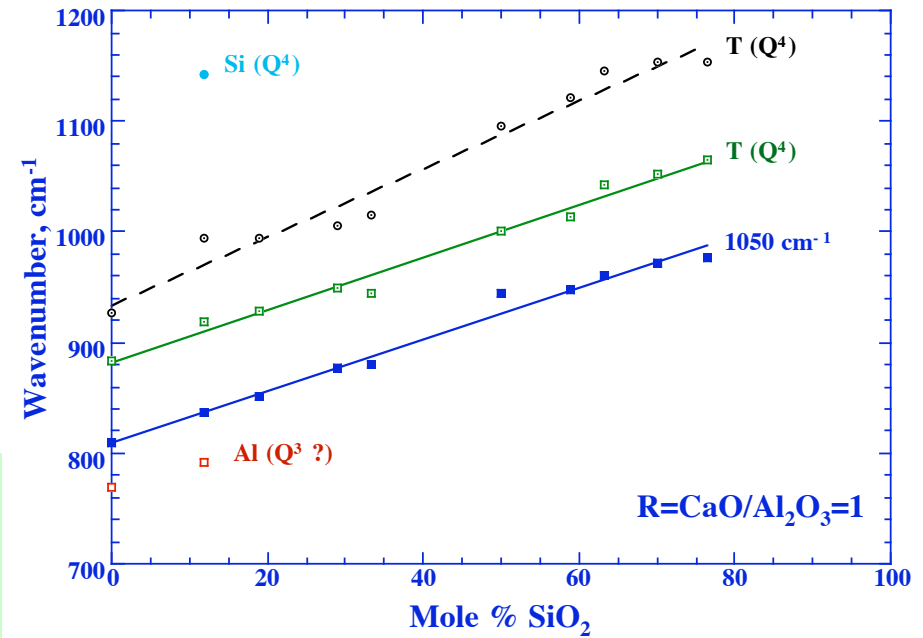
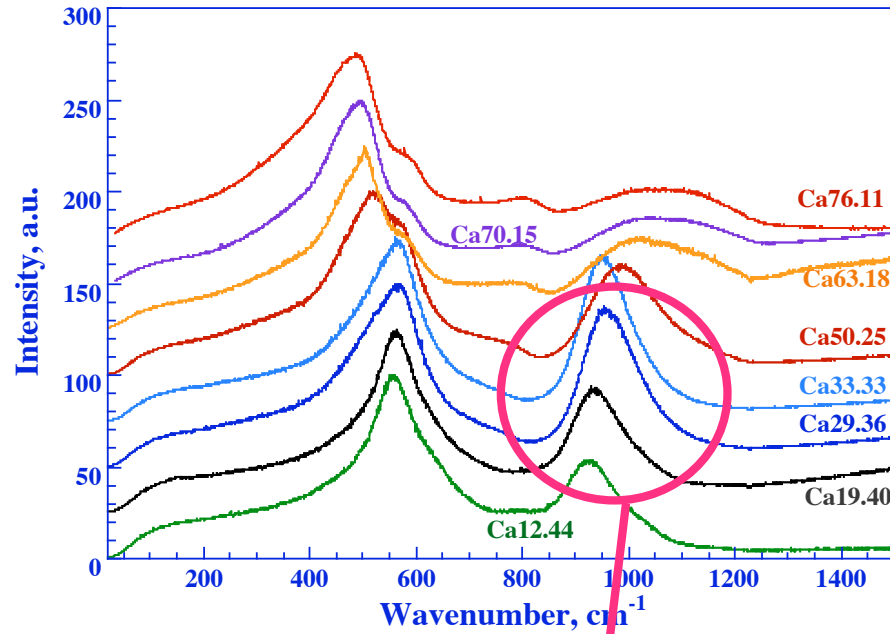
SiO₂
T_g=1400K

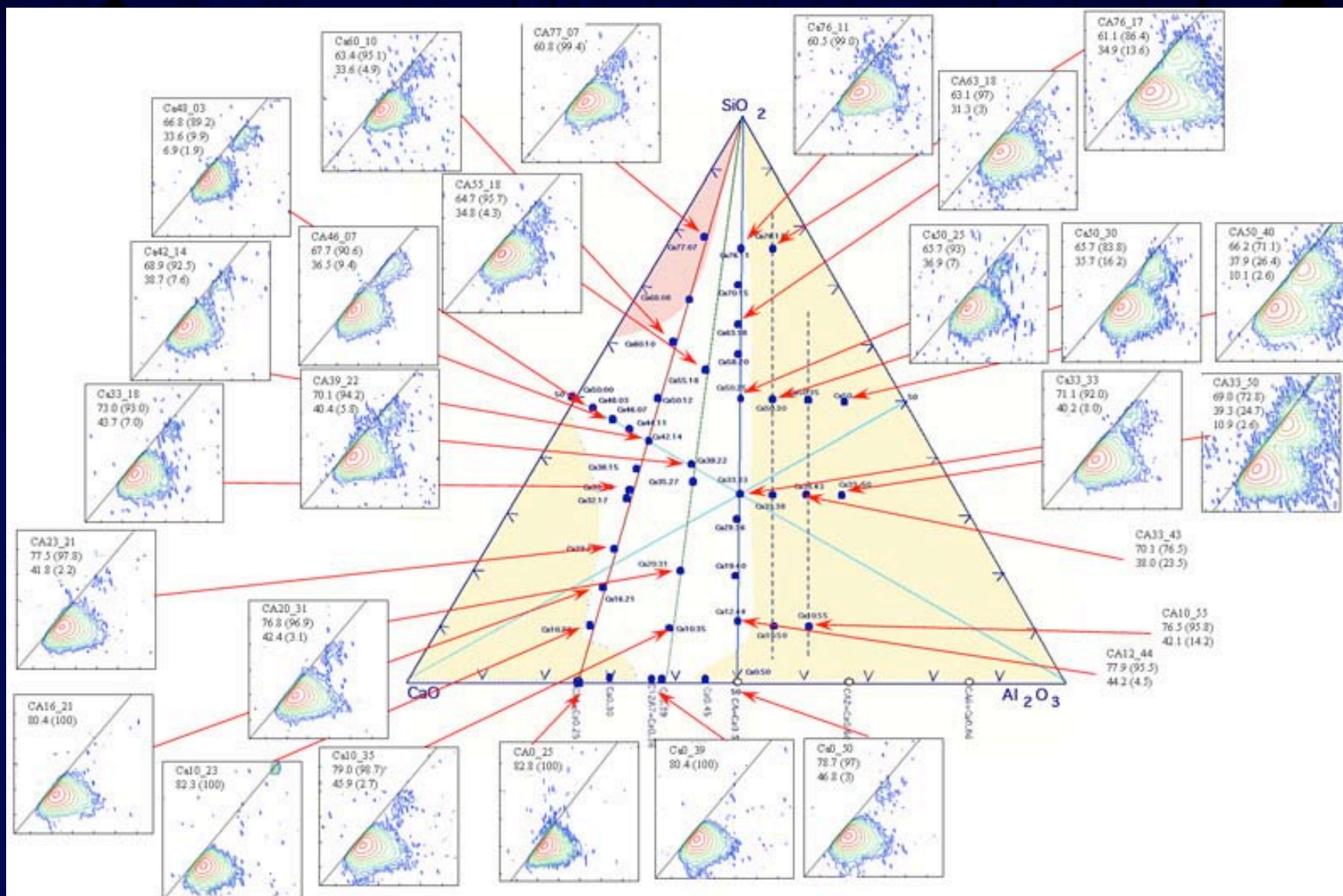
SiO₂ => Tetrahedra SiO₄
 CaAl₂O₄ => Tetrahedra AlO₄
 substitution of 1 Si by 1 Al and
 Ca charge compensator

Anomaly ?

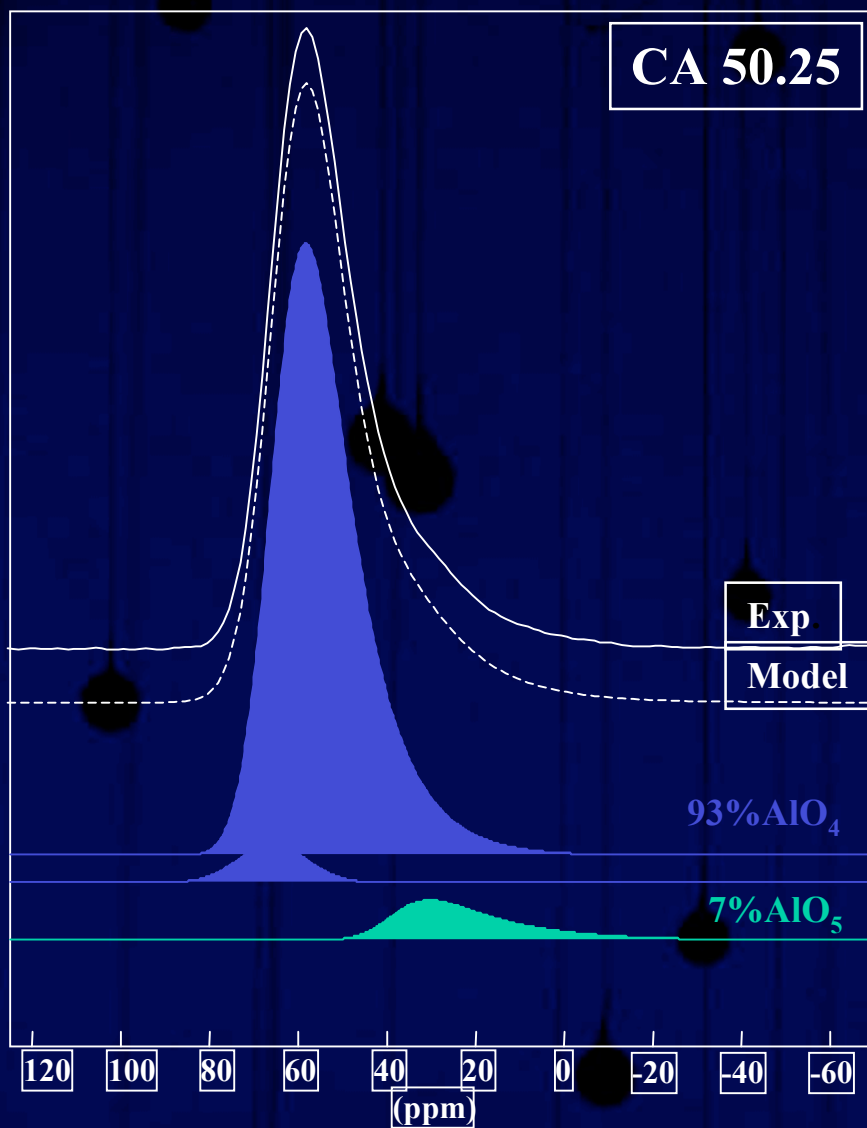


Decrease of CAS wavenumbers with SiO₂

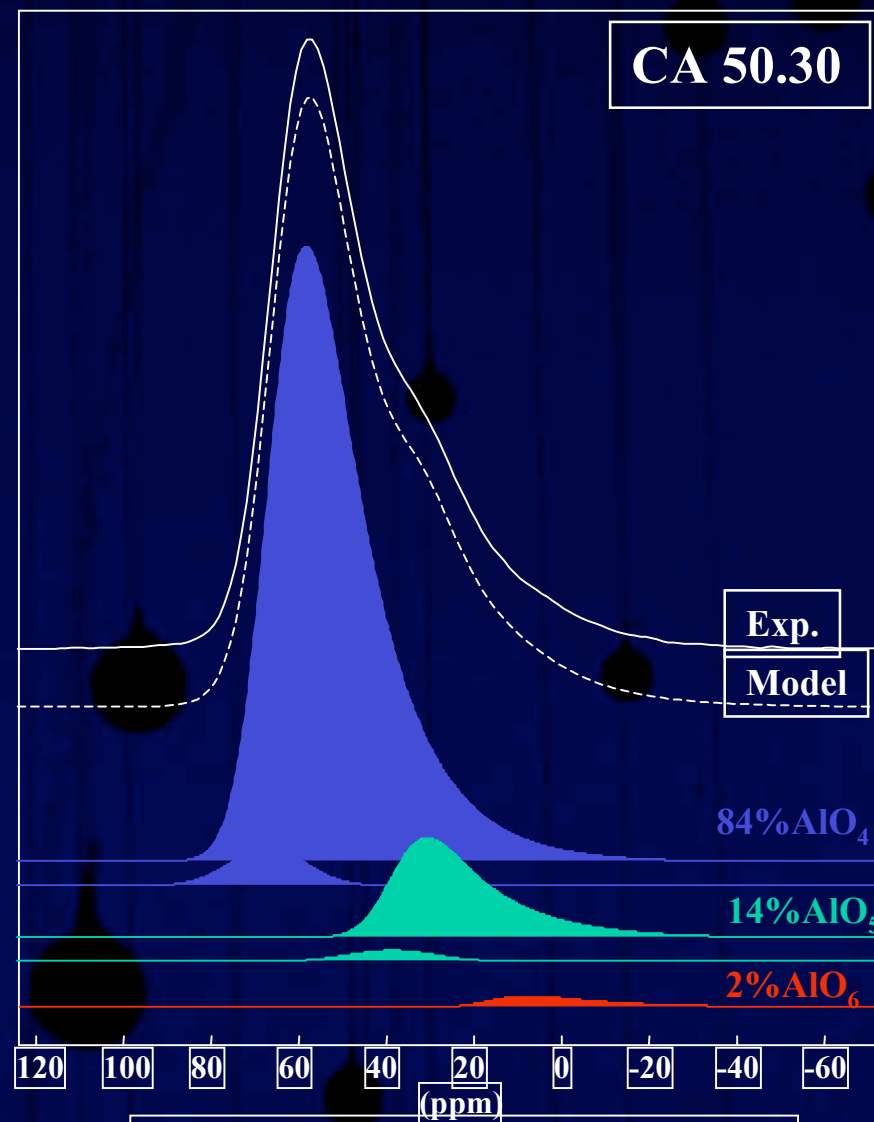




RMN 750MHz, CRMHT, Orléans, ^{27}Al 1D MAS



$$R = \text{CaO}/\text{Al}_2\text{O}_3 = 1$$



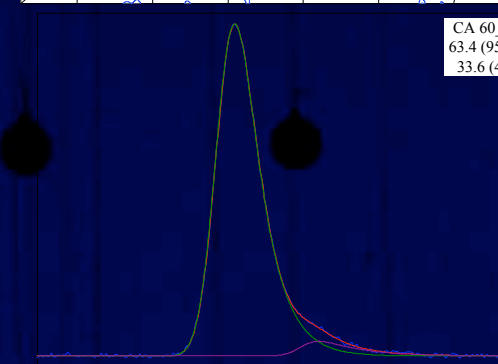
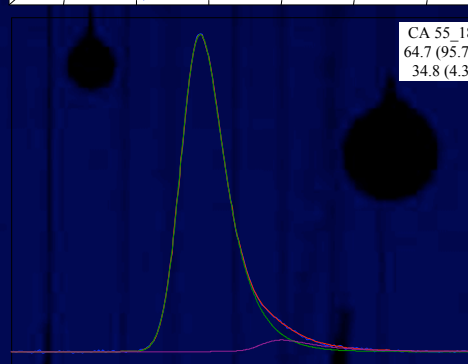
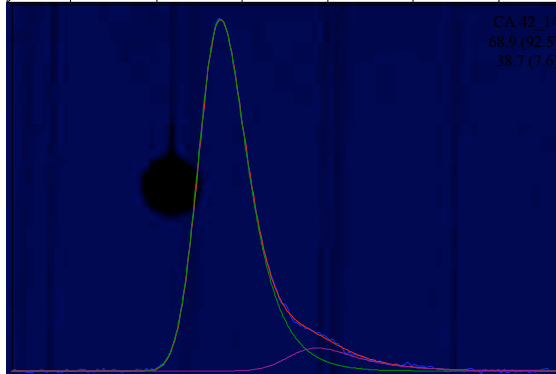
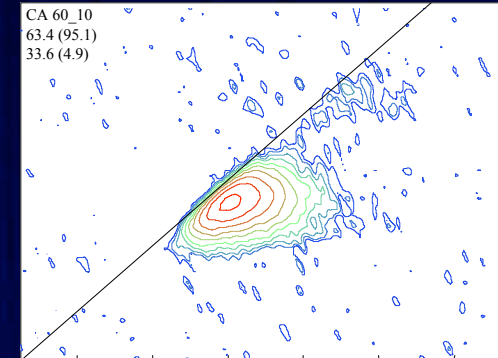
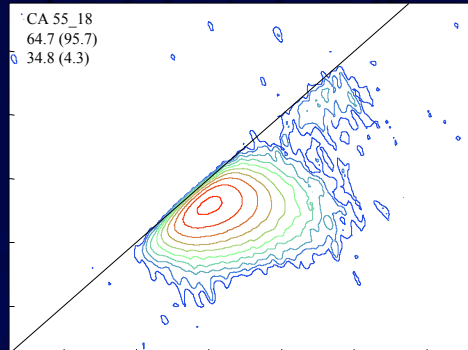
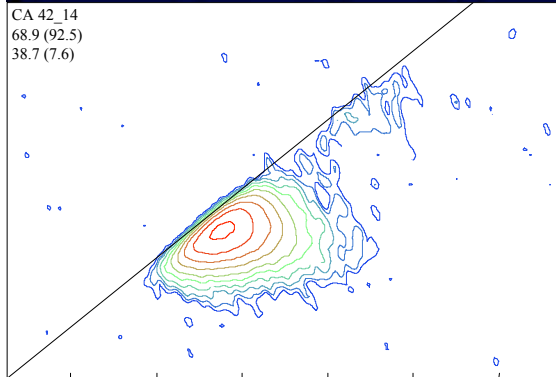
$$R = \text{CaO}/\text{Al}_2\text{O}_3 = 1 + \text{Al}_2\text{O}_3 \text{ ex}$$

$$R = \text{CaO}/\text{Al}_2\text{O}_3 = 3$$

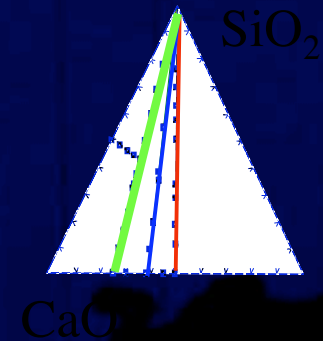
CA42.14

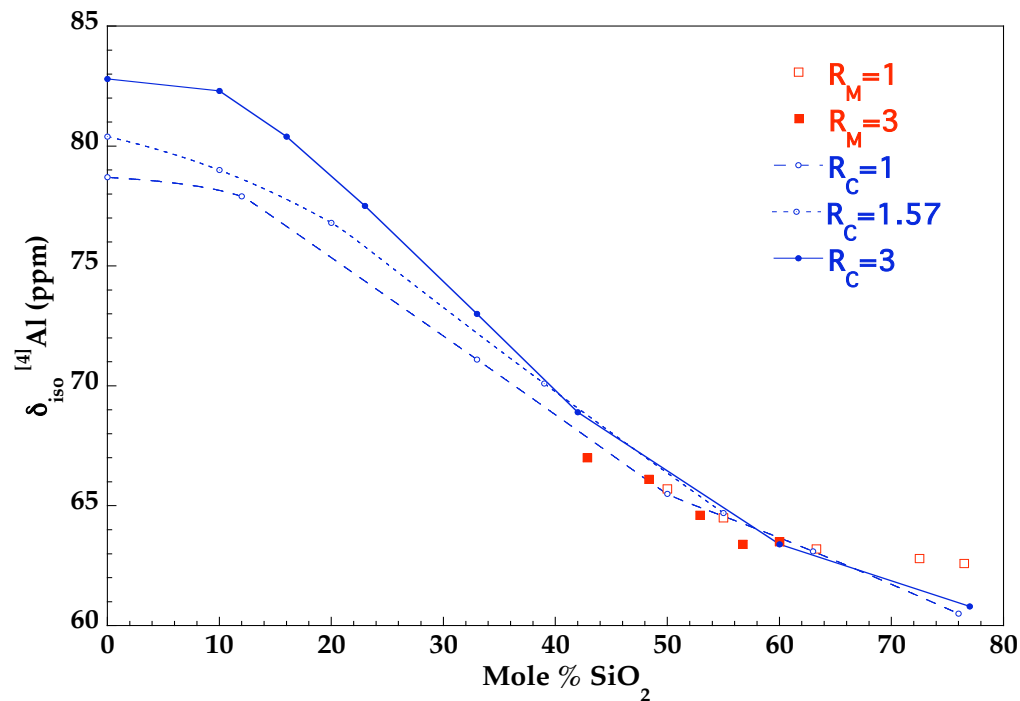
CA33.17

CA60.10

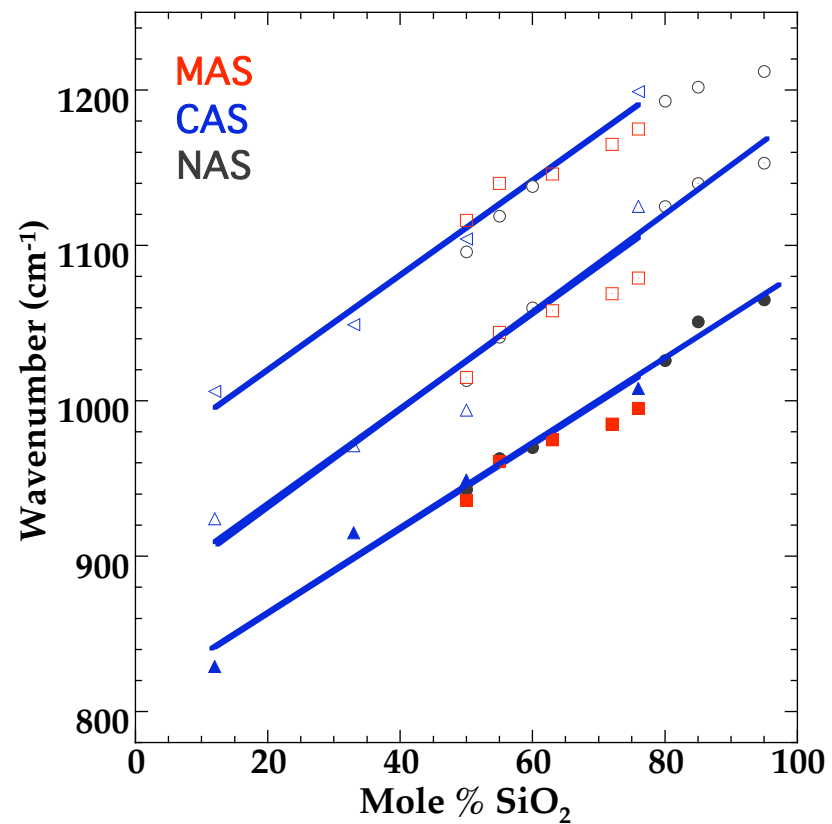


$\Rightarrow 1 < R < 3$ 92% Al^{IV} and 8% Al^{V} in glasses
with classic (15°/s) and rapid quench (300°/s)



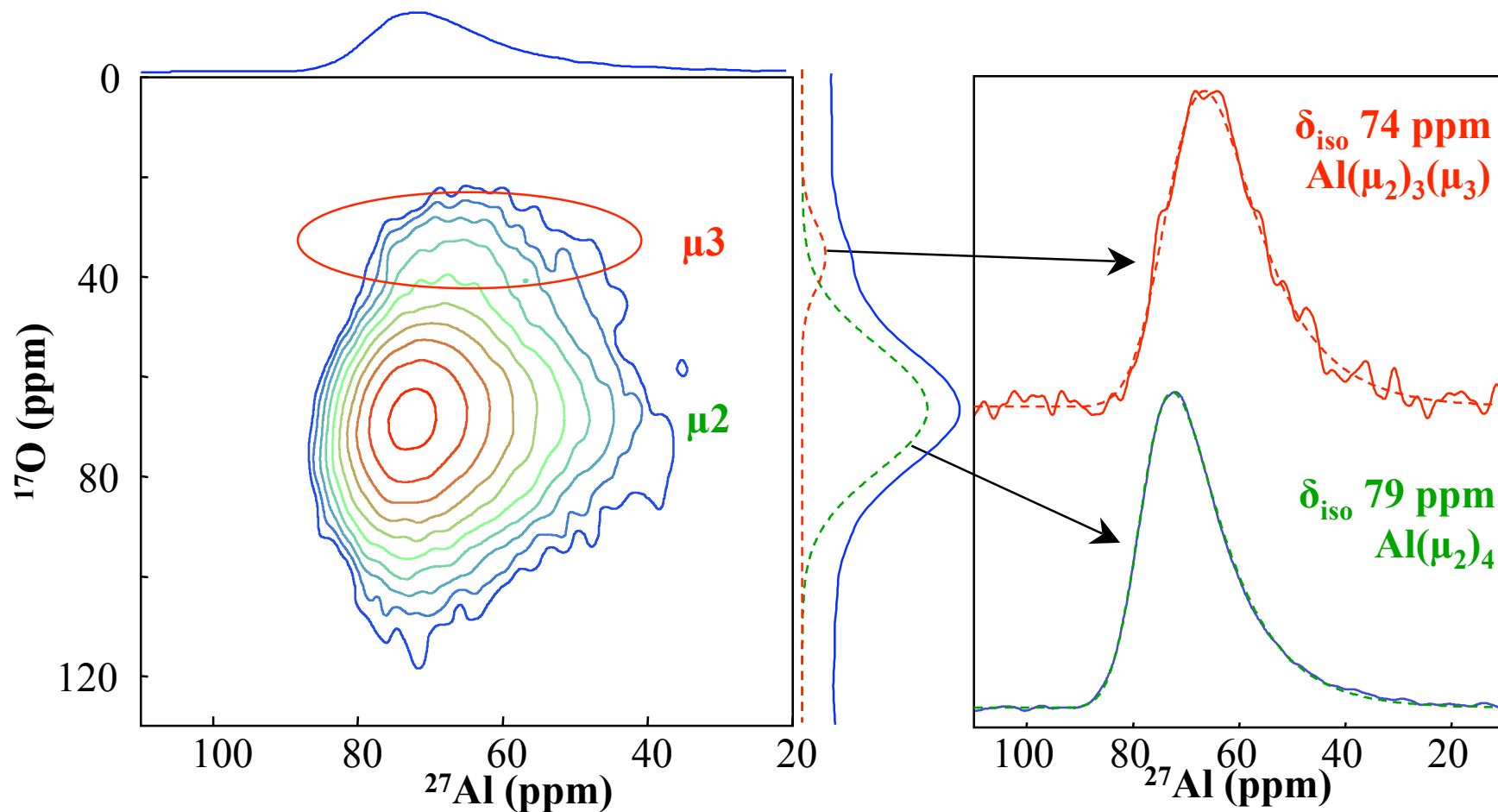


$\delta_{iso}^{[4]Al}$ independent of MO/Al_2O_3 and vary linearly with SiO_2

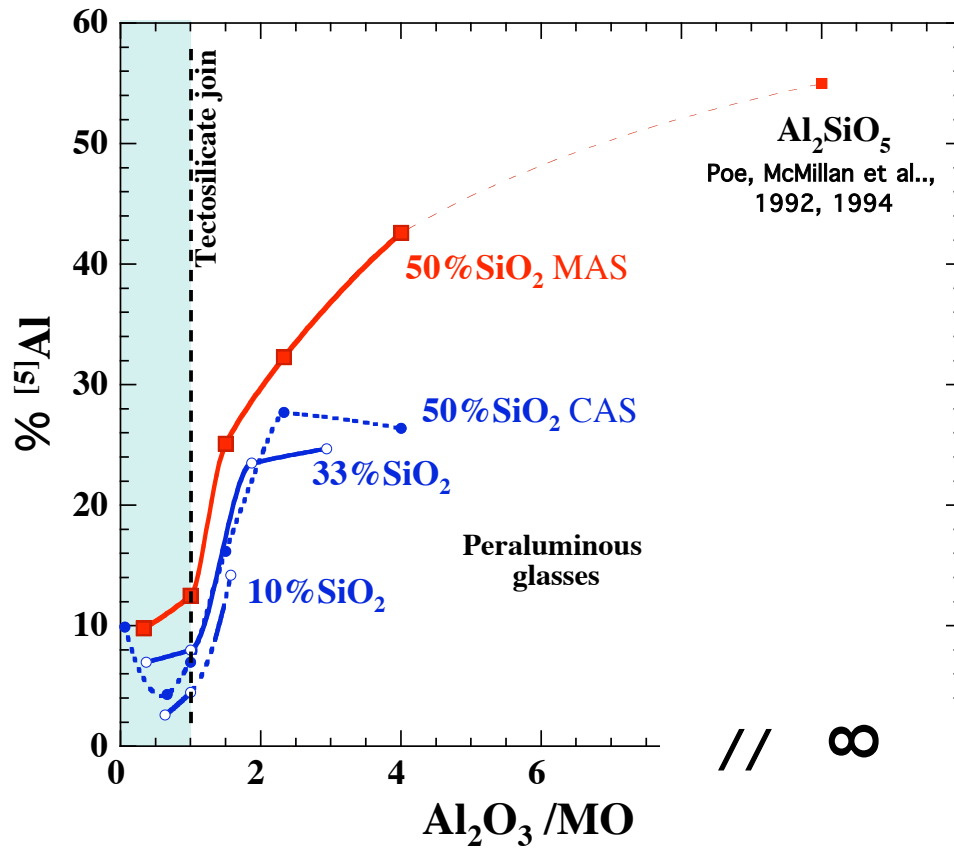


CaO Al₂O₃ - Glass

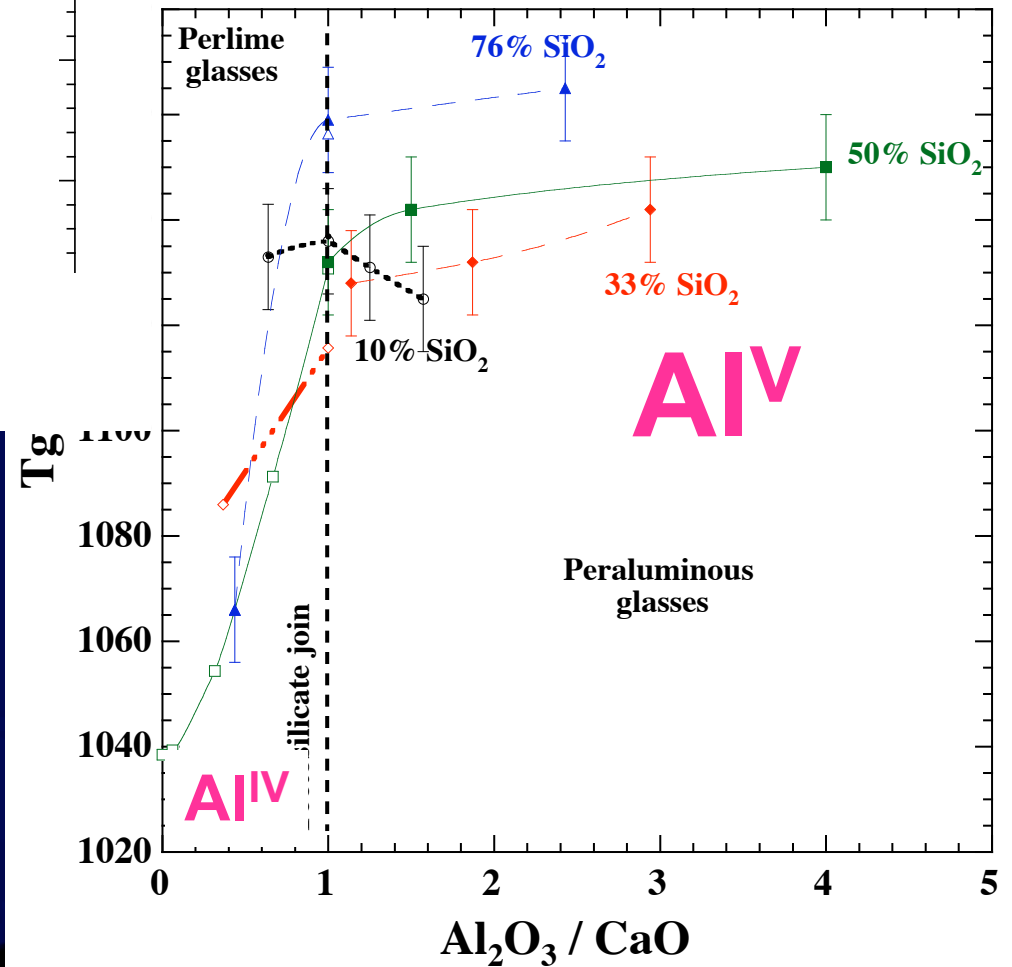
Hetero-nuclear correlation



Consistent ¹⁷O and ²⁷Al signature of Al(μ_2)₃(μ_3) structural entities



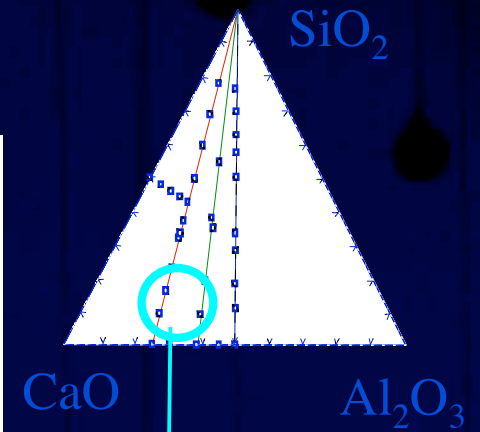
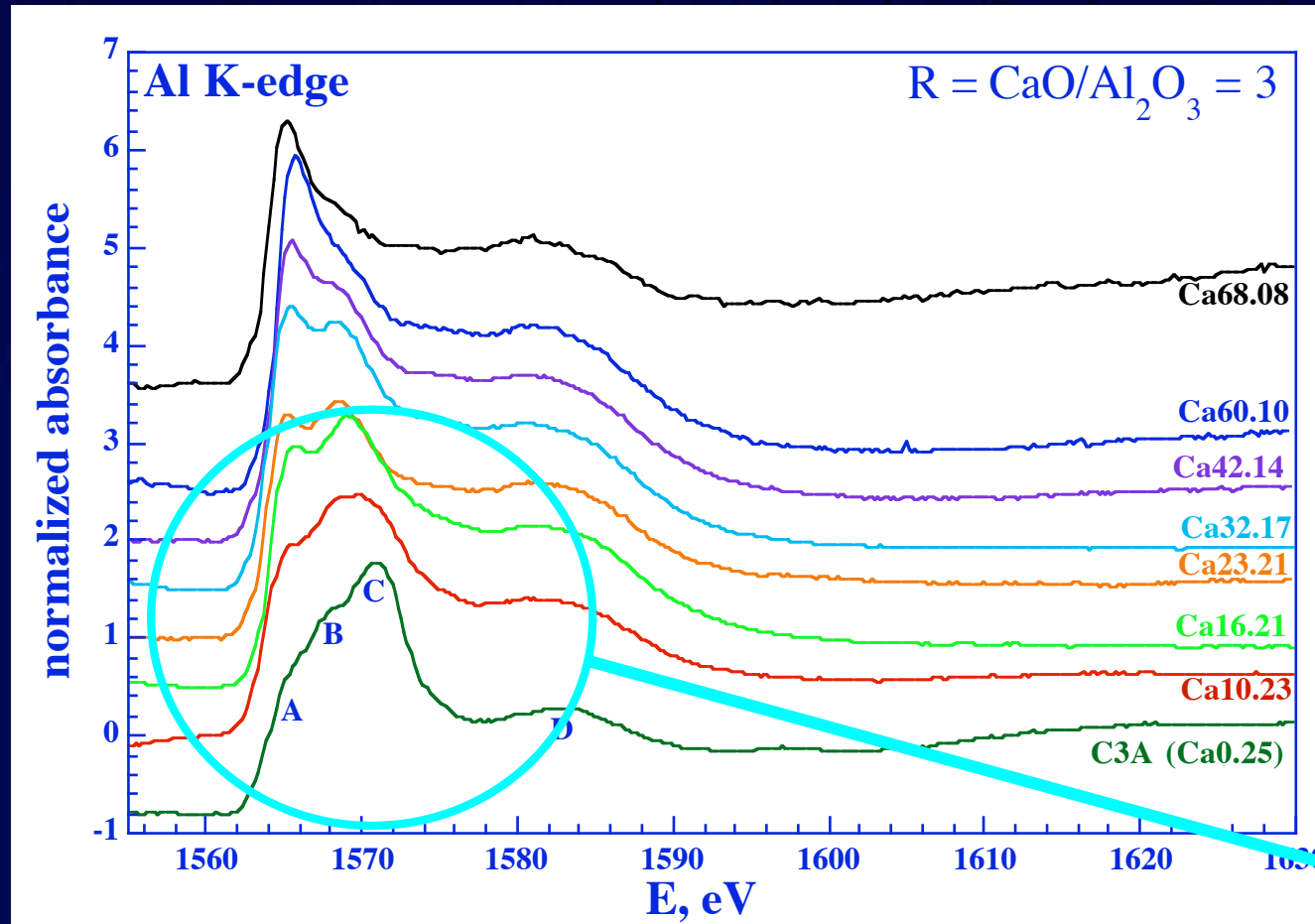
Al^V increases with Al₂O₃ content for R=CaO/Al₂O₃<1



Al^V increases the viscosity
Al^V has a role of network former

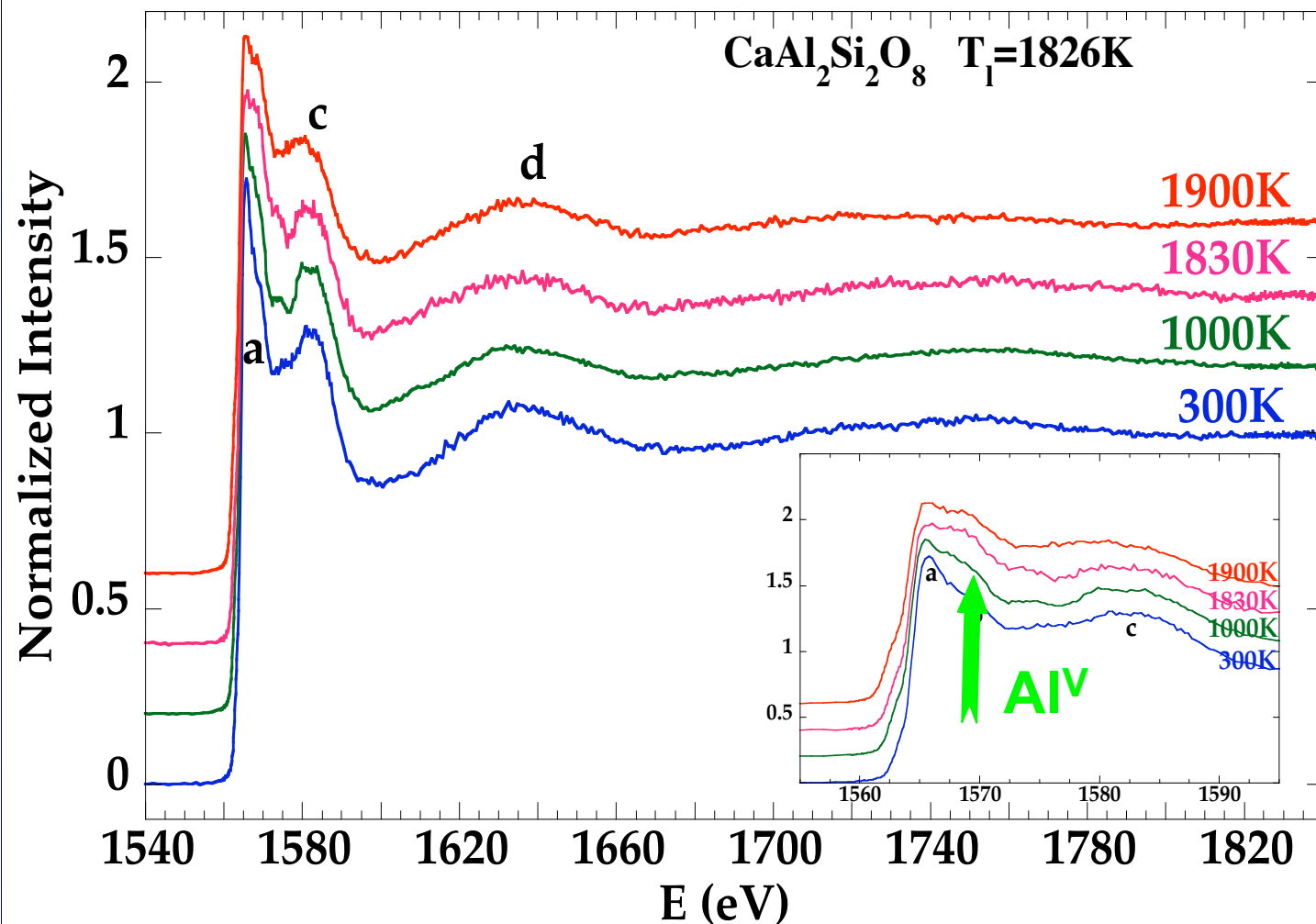
CaO-Al₂O₃-SiO₂ glasses: Al K-edge

SA32 SUPERACO- LURE



Al in 4-fold coordination

Al in
Q² species

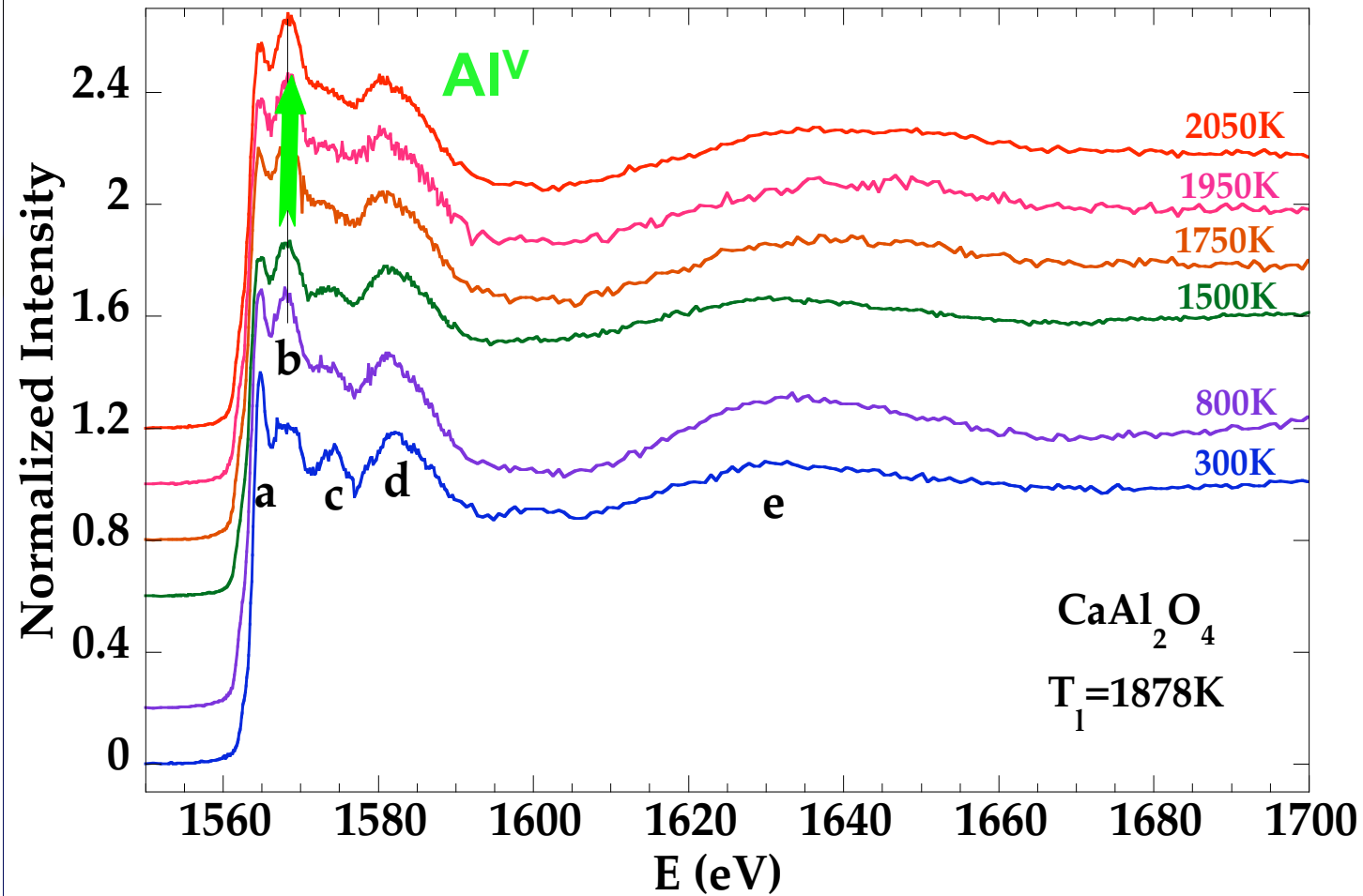


Al K-edge

Anorthite
 $\text{CaAl}_2\text{Si}_2\text{O}_8$

Crystal and melt 1000K => Al in 4 fold coordination

with increasing temperature few Al in 5 fold coordination appear
according with NMR (Coté, 1993) and Raman spectroscopy (Daniel et al, 1995)

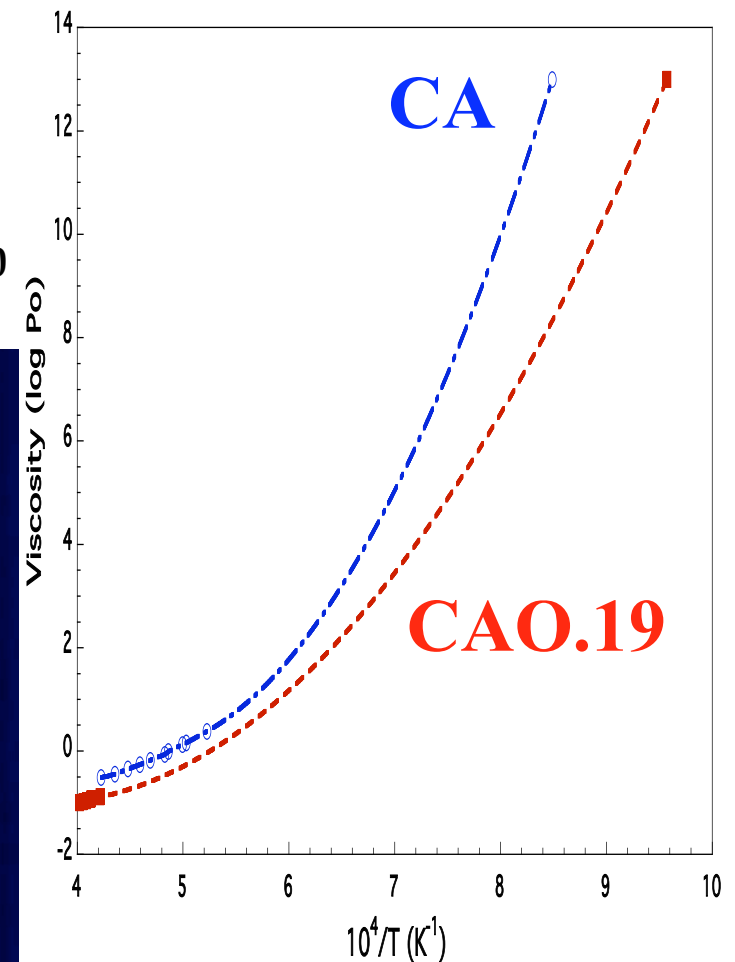
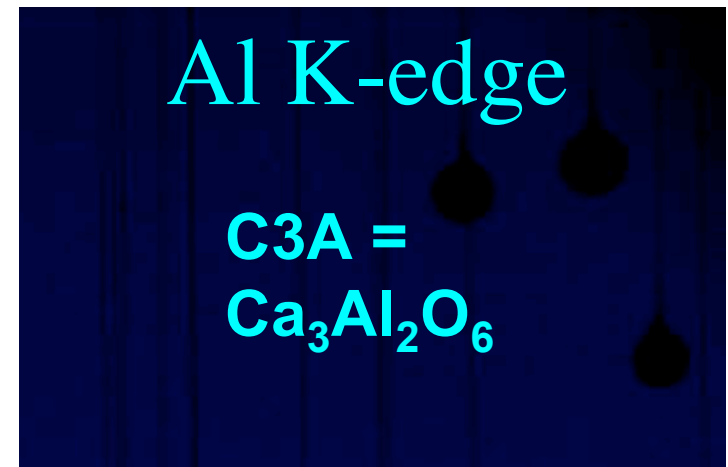
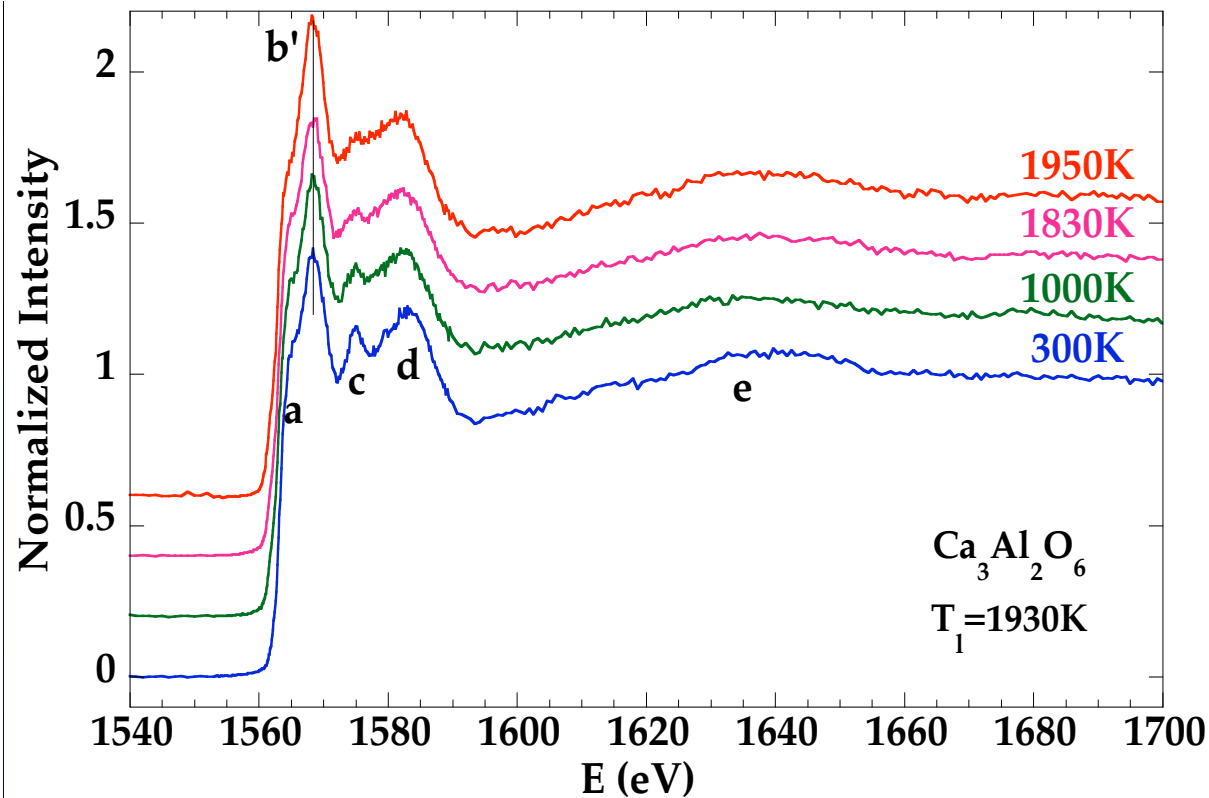


Al K-edge

CA =
CaAl₂O₄

Crystal and melt => Al in 4 fold coordination

with increasing temperature few Al in 5 fold coordination appear
according with NMR (Couture et al, 1990)



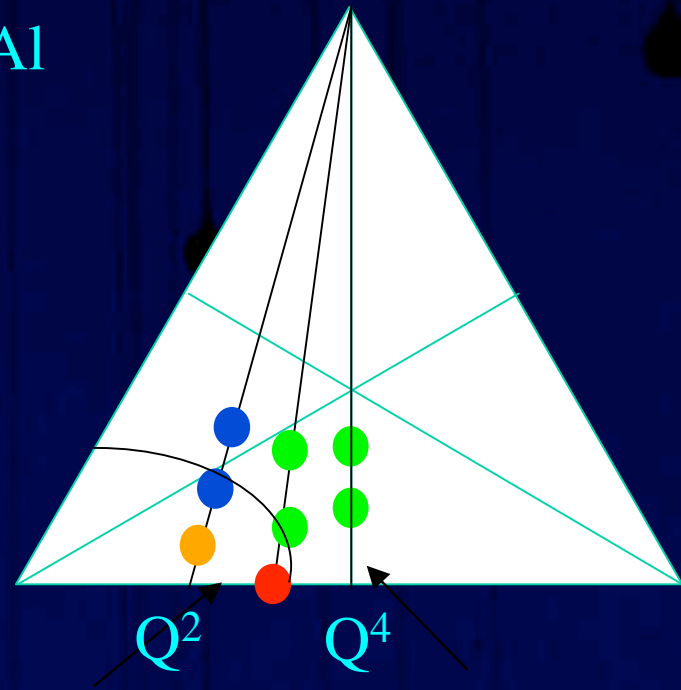
Crystal => Al in 4 fold coordination with 2 Bridging Oxygen, Q^2 species with increasing temperature no changes are observed

Viscosity decrease with CaO, and Al evolves from Q^4 to Q^2 with CaO

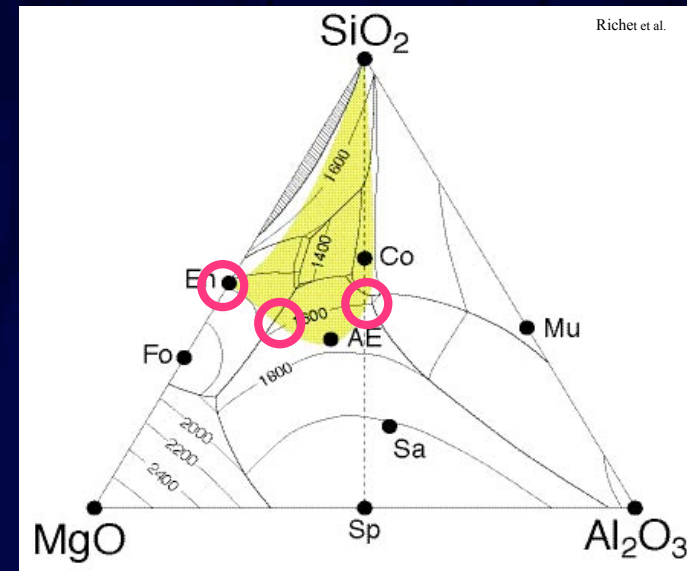
Ca-Conclusions

Explanations for the increase of T_g at low SiO_2 content

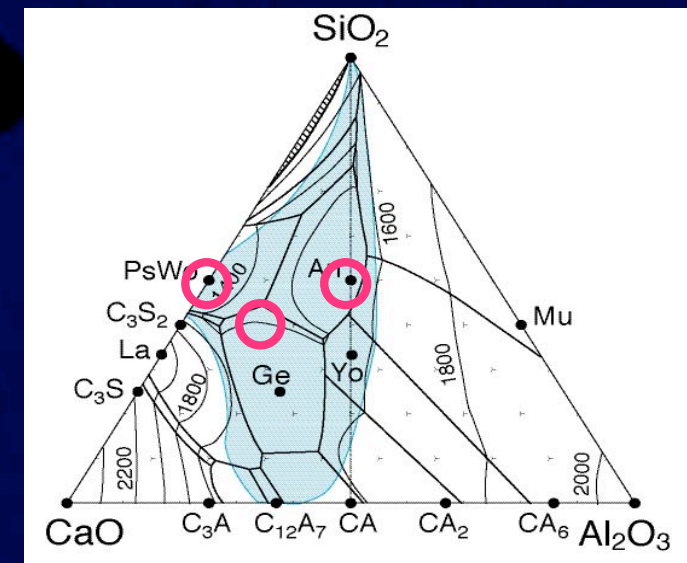
- Glasses $R=1$: Q^4
few structural change - substitution Si/Al
 \Rightarrow polymerization not change
 \Rightarrow no T_g maximum
 \Rightarrow $^{[5]}\text{Al}$ explain the T_g deviation
- High content in CaO : Al in Q^2 low T_g
- With increase of SiO_2 or Al_2O_3 : Q^4
Al enters preferentially in Q^4 species
 \Rightarrow the connectivity of the network is increases \Rightarrow higher viscosity
 \Rightarrow high T_g
- Not need O tricluster to explain viscosity variation



1) MAS System

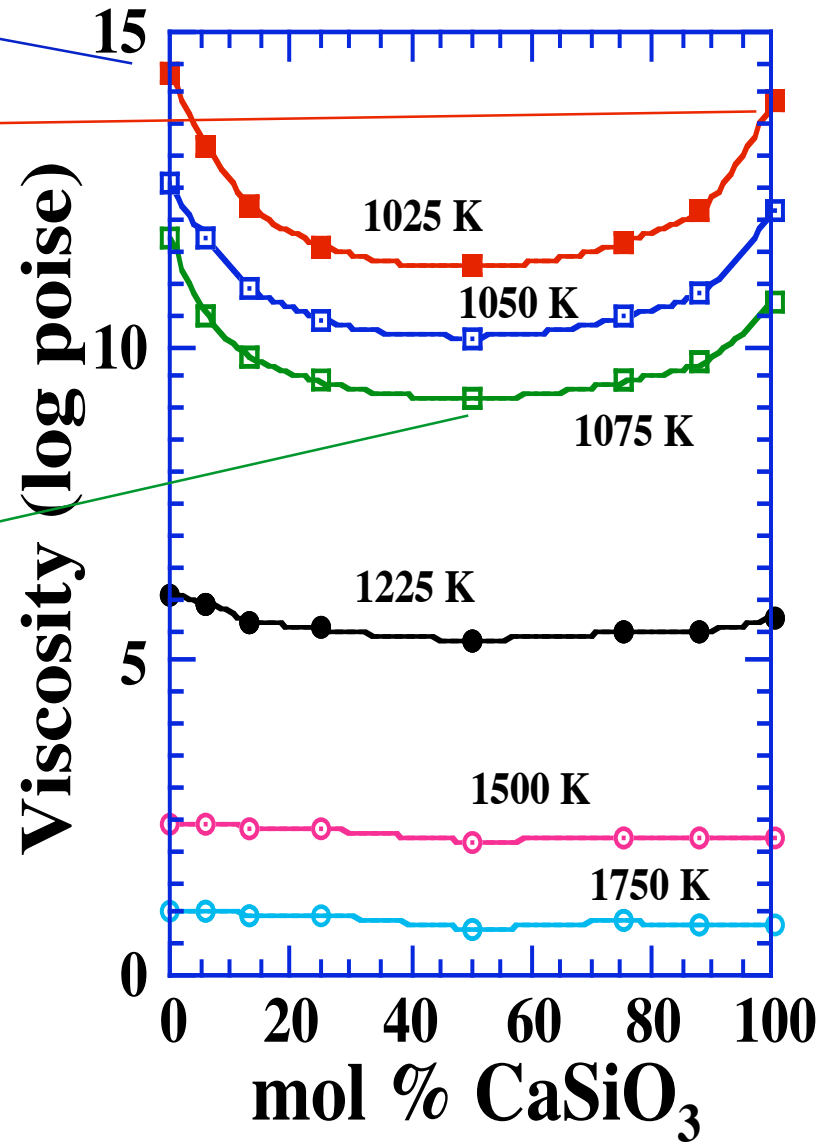
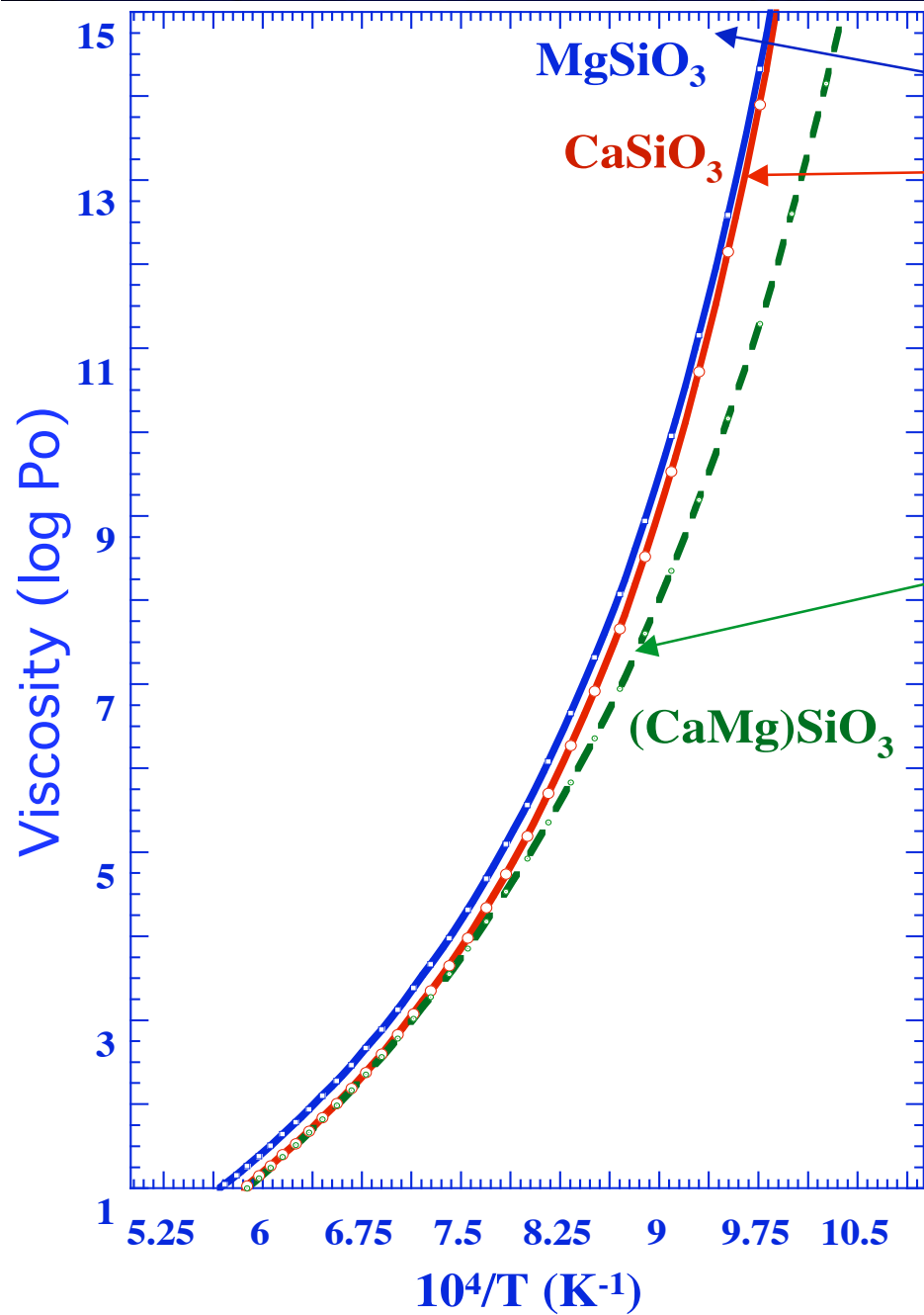


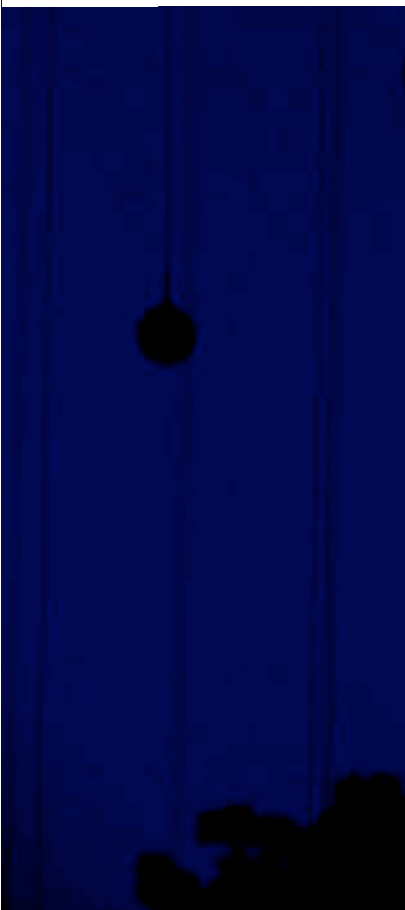
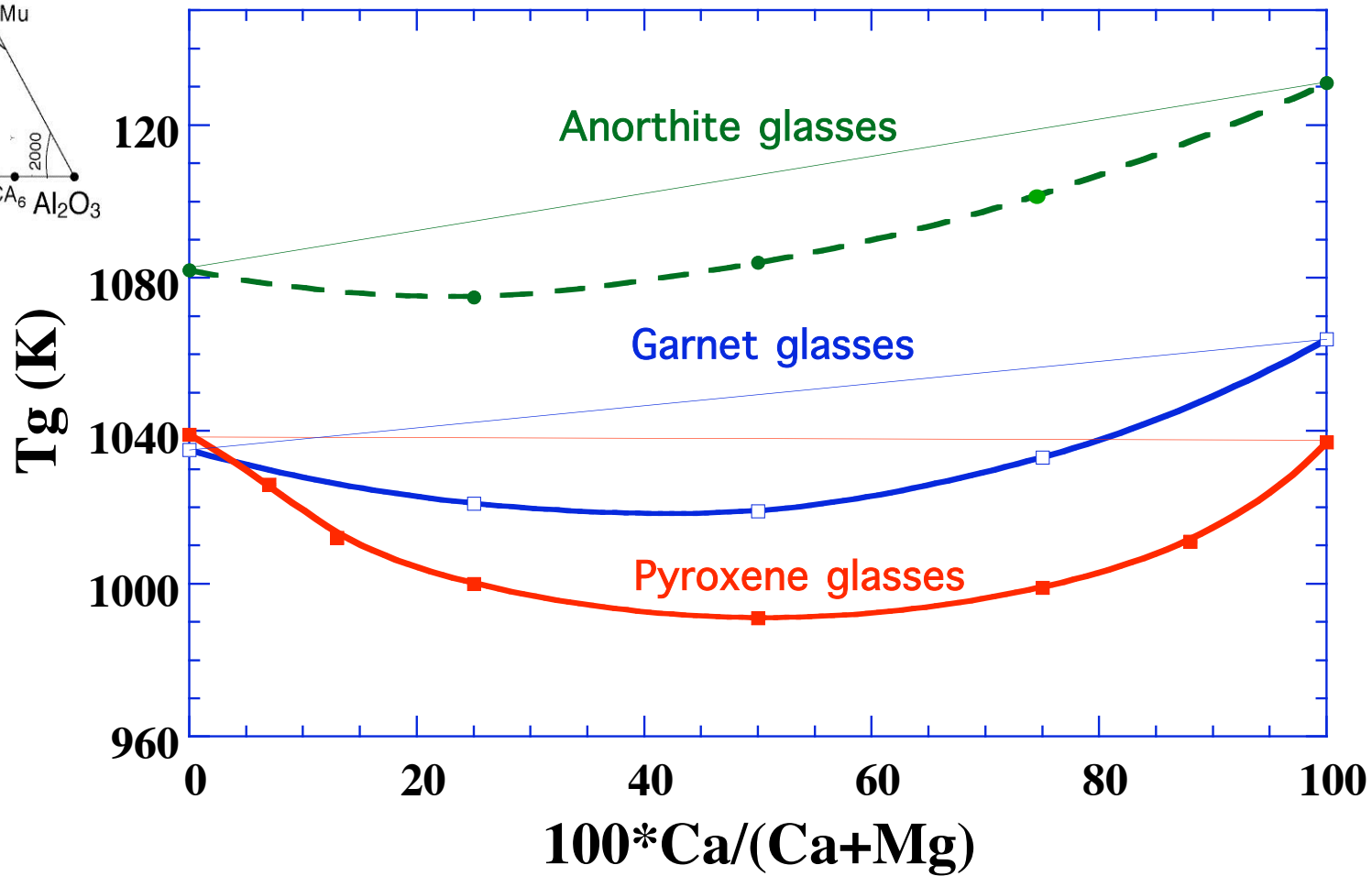
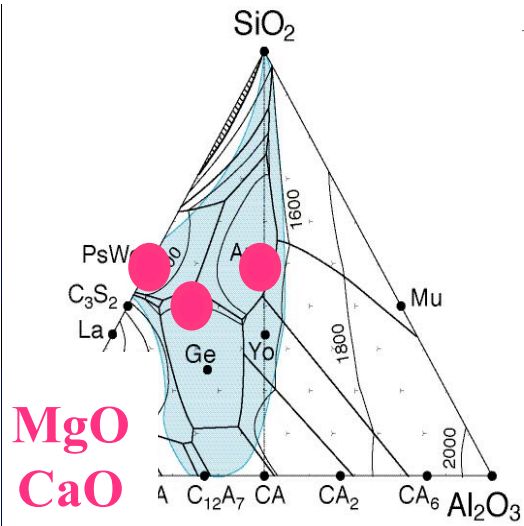
3) CMAS System

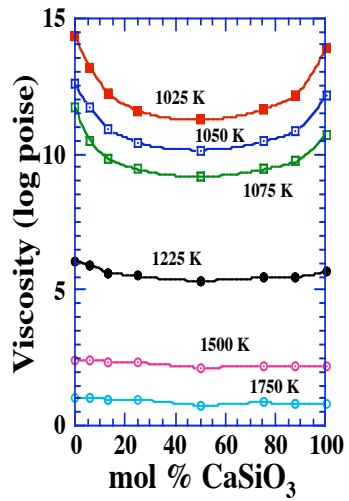


2) CAS System

Mixing Ca/Mg







$$\log \eta = A_e + B_e/TS^{conf}(T)$$

$$S^{conf}(T_g) = S^{mix} + \sum X_i S_i^{conf}(T_g)$$

$$S^{conf}_{top} = \sum X_i S_i^{conf}(T_g)$$

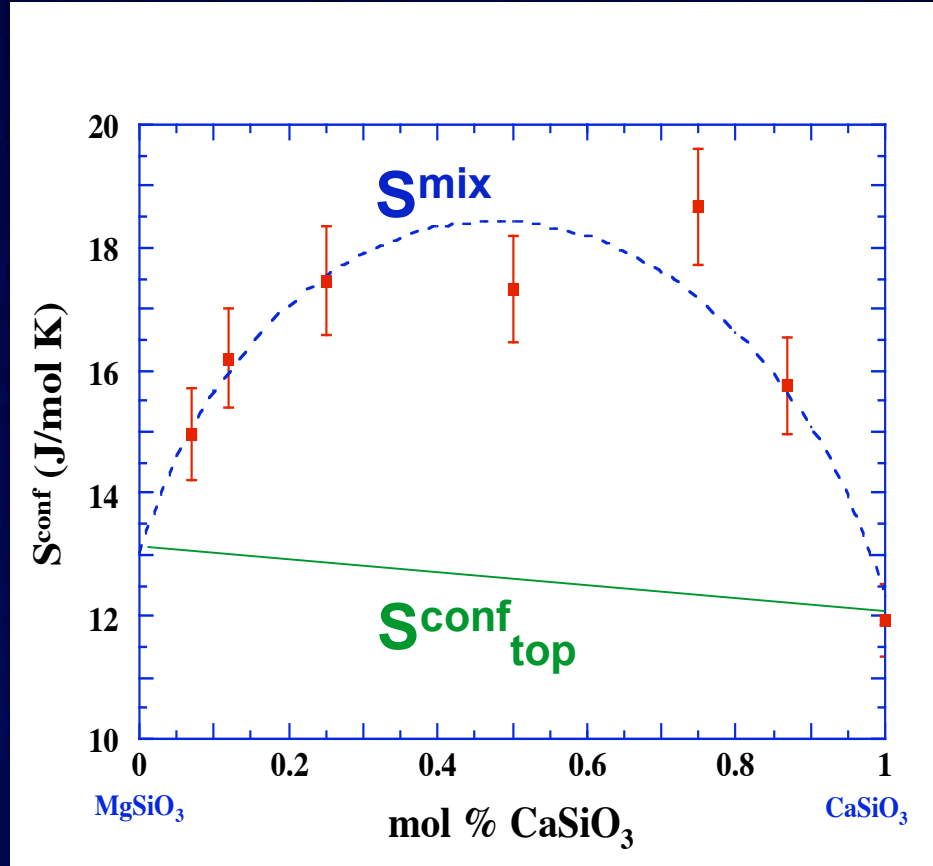
$$S^{mix} = -nR \sum X_i \ln X_i$$

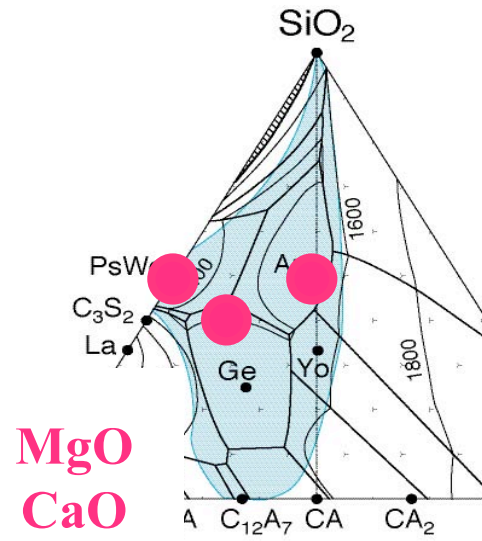
$$X_i = Ca/(Ca+Mg)$$

Ideal mixing => random distribution

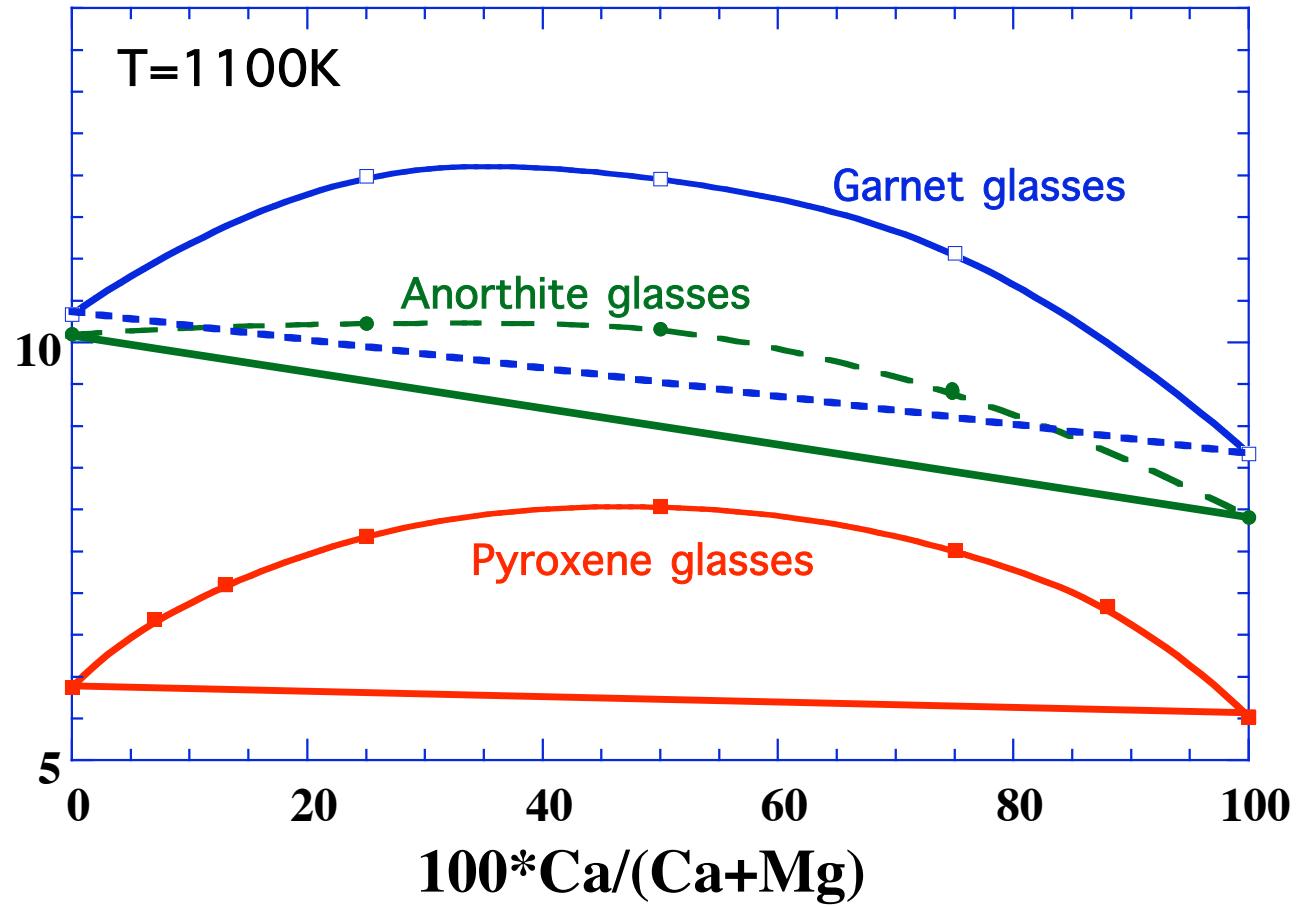
$$S^{conf}(T) = S^{conf}(T_g) + \int_{T_g}^T C_p^{conf}/T dT$$

$$with C_p^{conf} = C_p^l - C_{pg}(T_g)$$





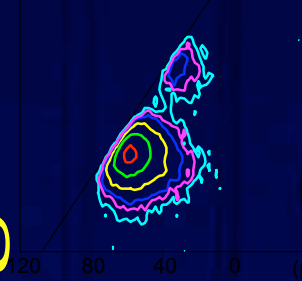
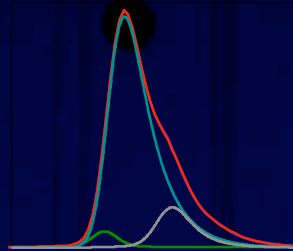
S_{conf} (J/Mol K)



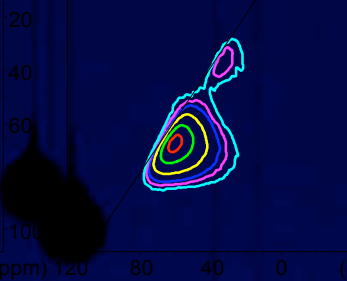
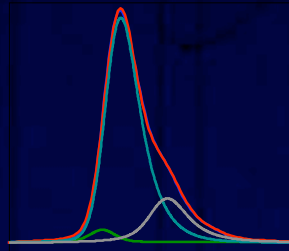
1 unit formula

Anorthite glasses

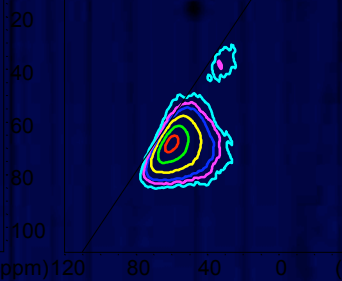
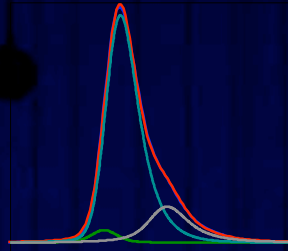
CMA50.25.00



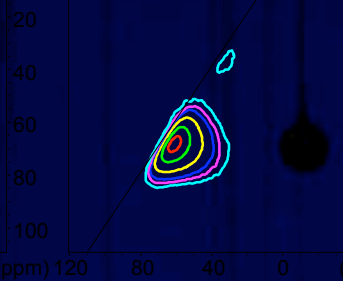
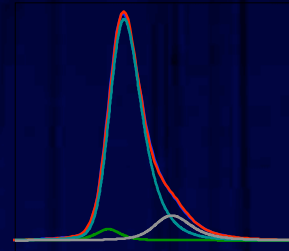
CMA50.25.6



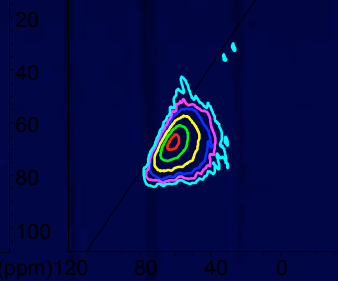
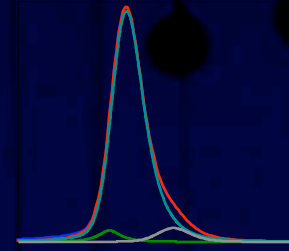
CMA50.25.12



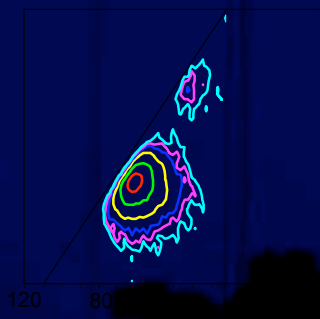
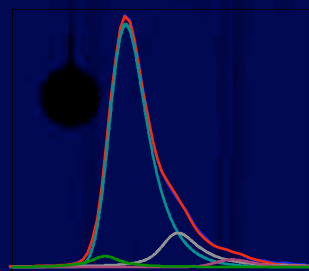
CMA50.25.18



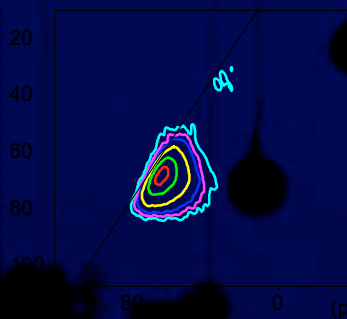
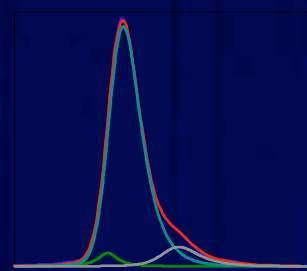
CMA50.25.25



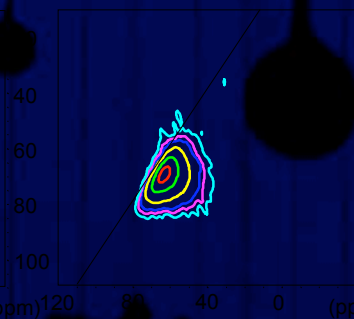
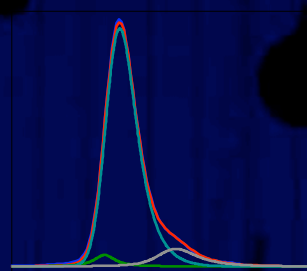
CMA42.14.00



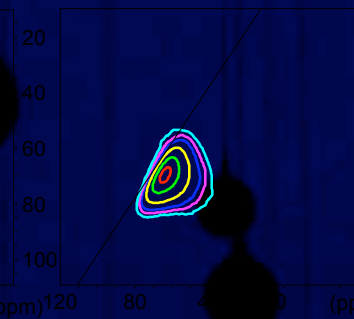
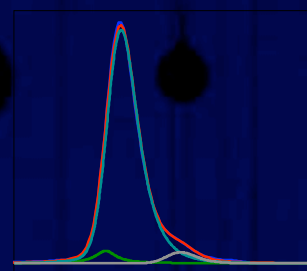
CMA42.14.10



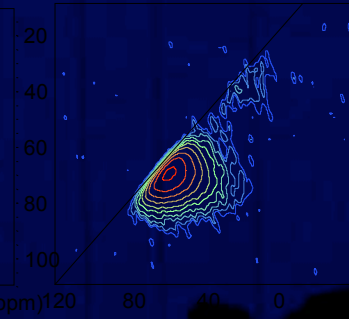
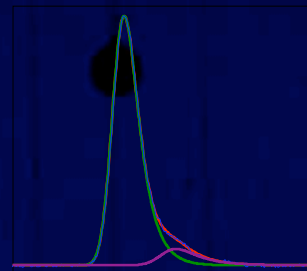
CMA42.14.21



CMA42.14.32



CMA42.14.42



MgO

CaO

Garnet glasses

Ca/Mg-Conclusions

- **No significant changes in Raman spectroscopy**
- **viscosity measurements show a minimum at T_g which can be explain by an ideal mixing term in the configurational entropy**
- **the proportion of $^{[5]}Al$ increases with Al_2O_3**

Conclusions

- **R=1** : substitution of Si by Al in Q⁴ species see by Raman, NMR are in good agreement with viscosity and configurational entropy
- per-MO glasses: low amount of ^[5]Al and for the **CAS** system, Al in Q² species for low SiO₂ content.
- peraluminous glasses: ^[5]Al increases with Al₂O₃
- T_g increases with ^[5]Al => ^[5]Al can be a network former
- **Ca/Mg** mixing => ^[5]Al increases with Mg and viscosity can be predict using an ideal mixing term
- No tricluster oxygen to explain properties variation in **MAS**, **CAS**, **CMAS** and probably also in NAS